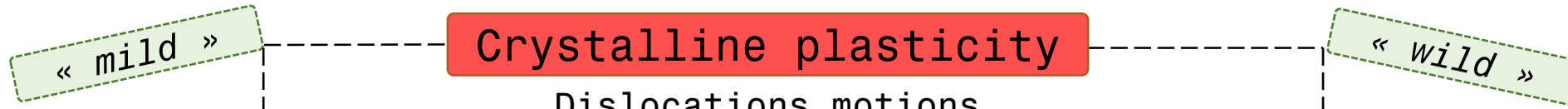




Study of dislocation dynamics under cycle fatigue : acoustic emission and microstructural characterizations

G.L'hôte, S.Cazottes, S.Deschanel

MATEIS, INSA Lyon, France

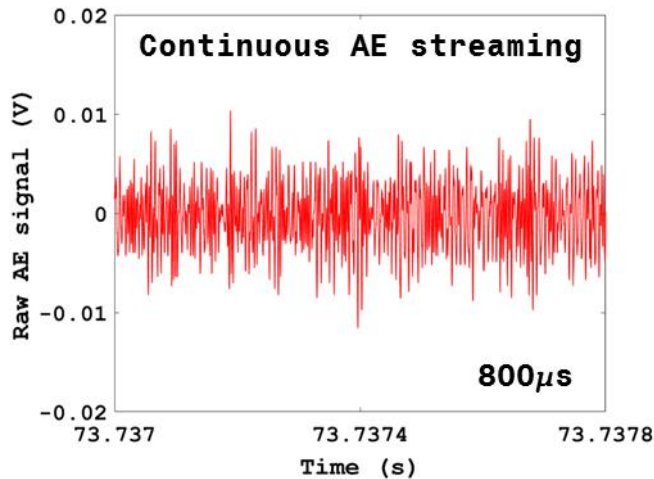


« Classical »
Homogeneous deformation in time and space

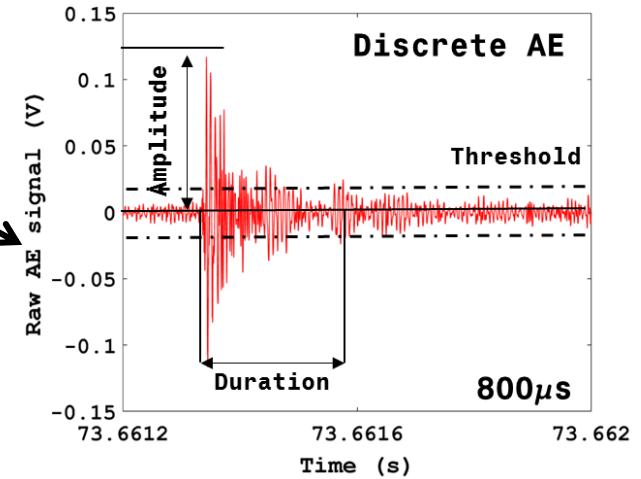
« Intermittent »
Heterogeneous deformation : dislocations avalanches

Numerous, smalls and uncorrelated motions of dislocations

Collective dislocation dynamics



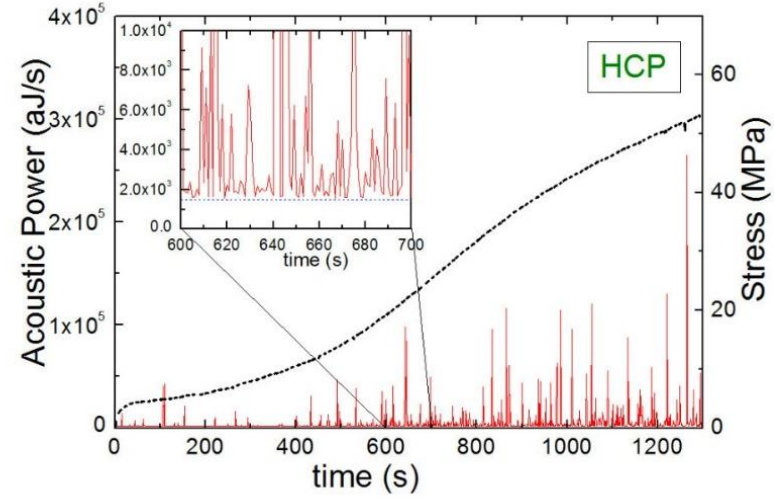
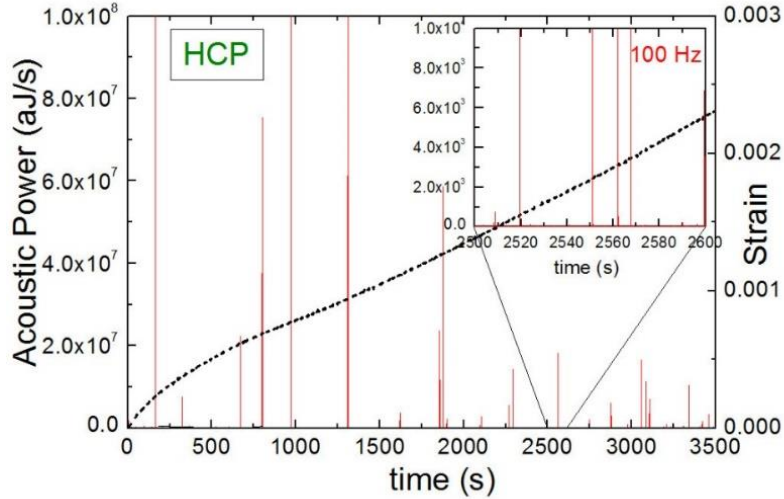
Acoustic Emission (AE)
A tool for exploring dislocation dynamics.



▪ **Continuous** : linked to the background noise of the signal (sampling rate : 1KHz)

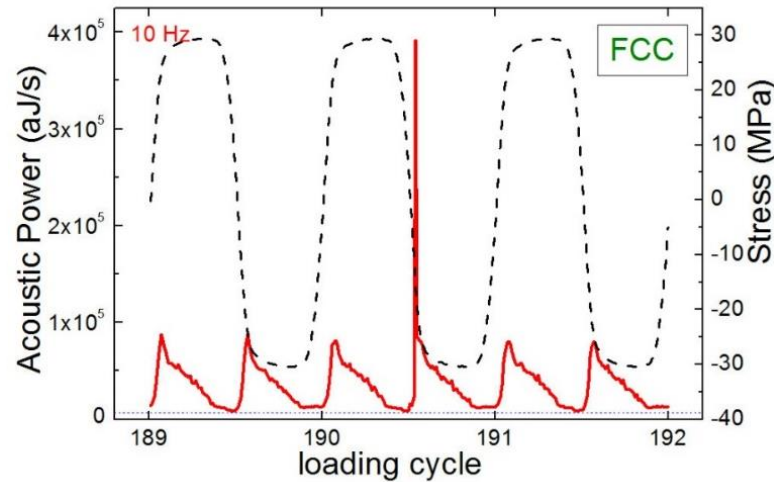
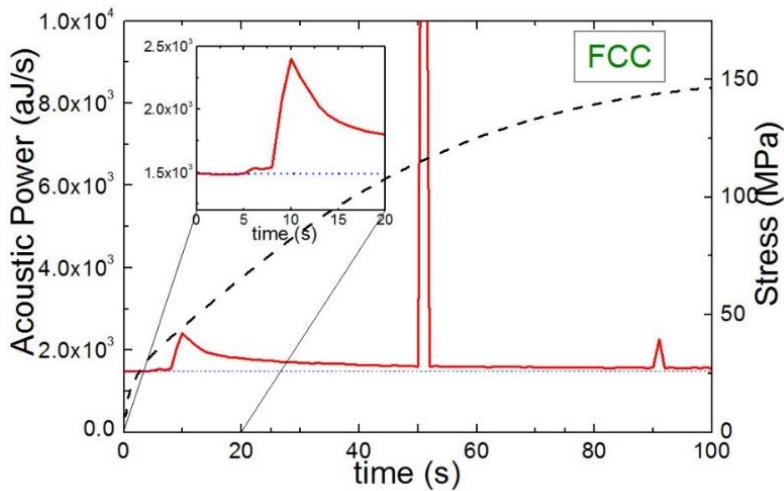
▪ **Discrete** : Define above a threshold

Ice
~ 100%



Cadmium
98 / 100%

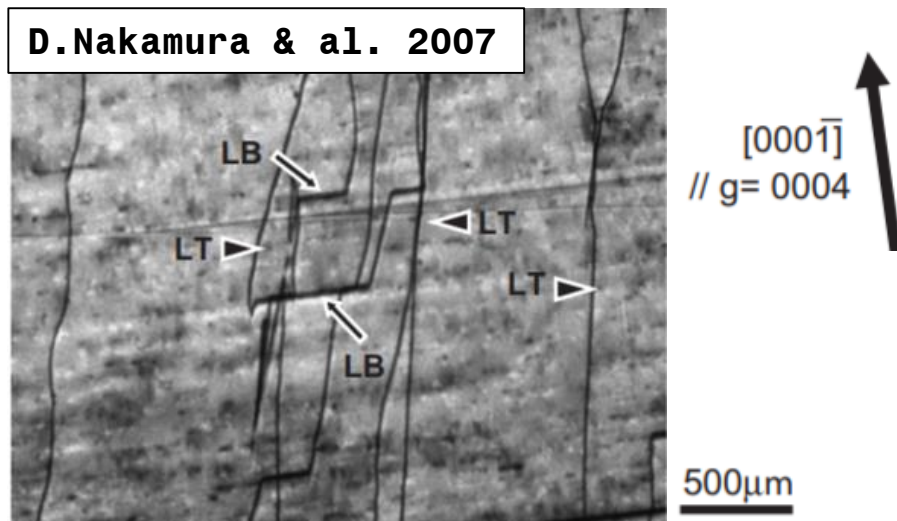
Copper
10 / 50%



Aluminum
<2%

Hexagonal (H/HC)

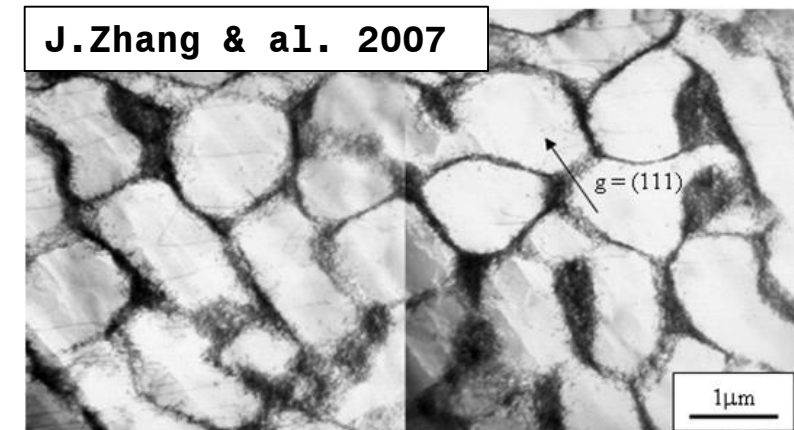
- Anisotropic glide (single slip)
- Long-range interactions between dislocations



Dislocation avalanches favoured

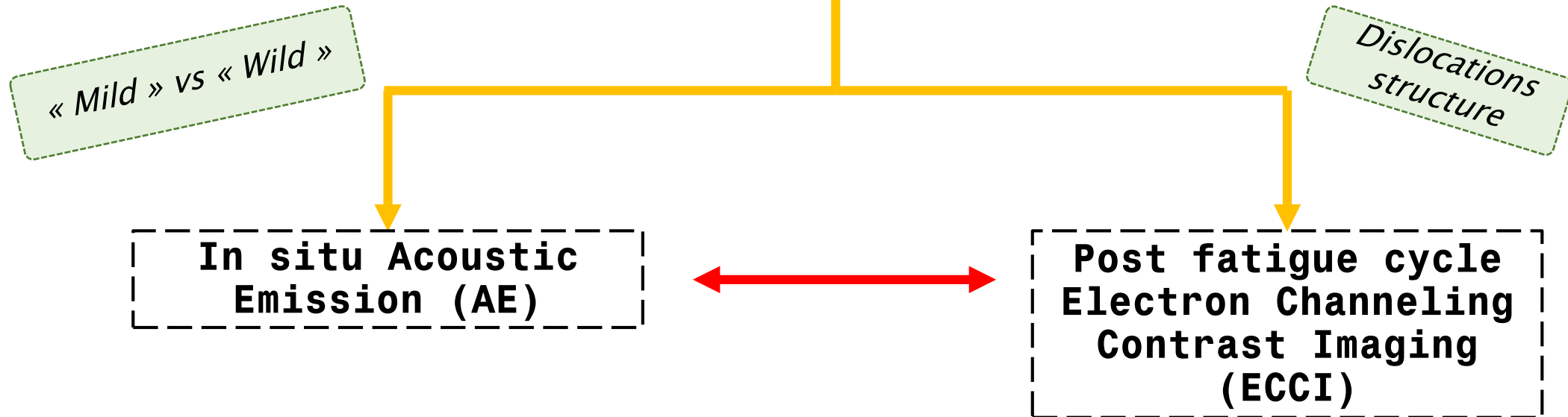
Face Centered Cubic (FCC)

- Isotropic glide (multiple-slip)
- Short-range interactions between dislocations
- Pattern formation



Mild plasticity favored & avalanches hindered

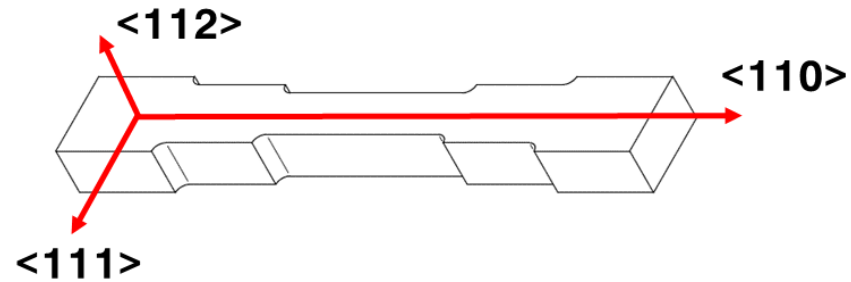
Dislocation dynamics of Cu single crystals under stress imposed fatigue test



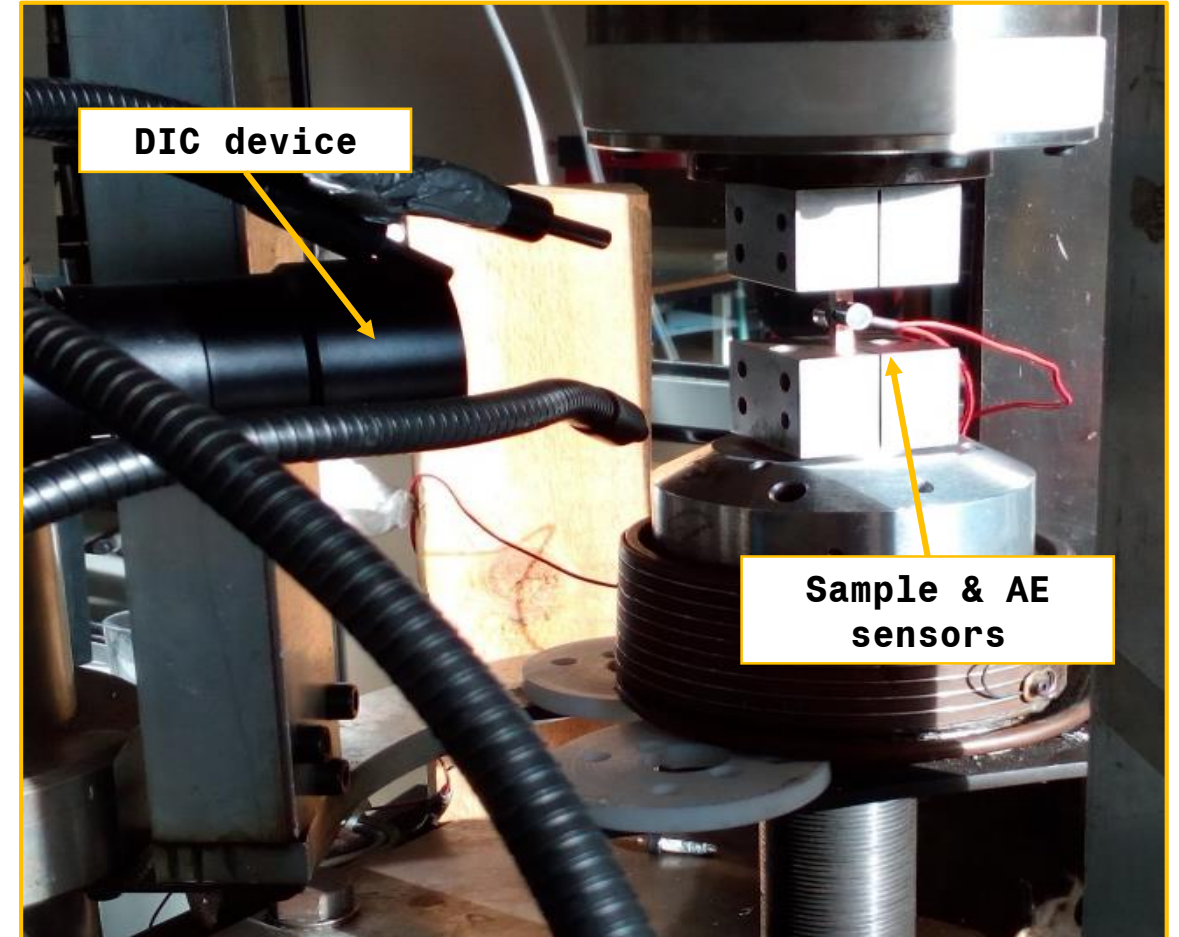
- Effect of crystallographic **orientation** on the dislocation dynamics.
- Comparison between **monotone** and **cyclic** experiment.

Copper single crystal (multiple slip)

(From P.Courtois (ILL, Grenoble))



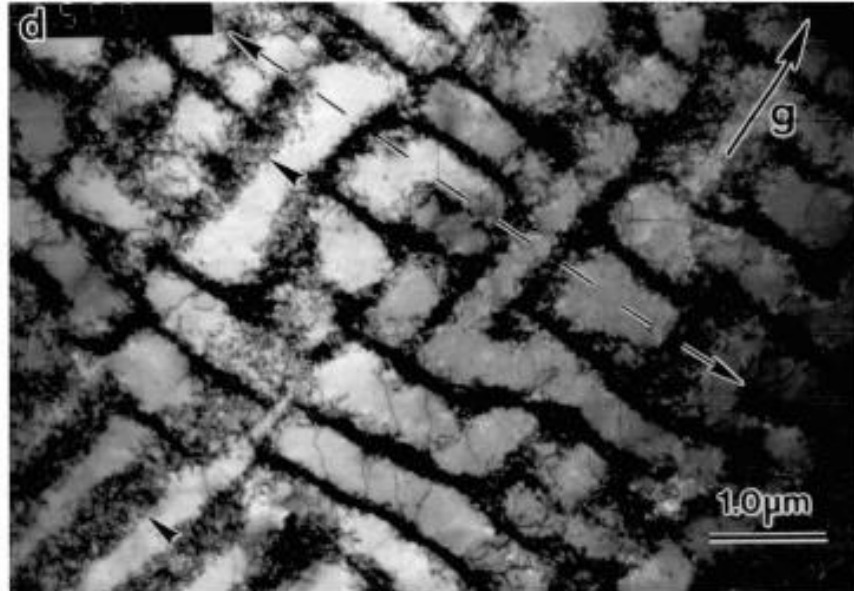
Fatigue machine	Frequency (Hz)	Loading mode
	0,1	Stress imposed
EA / DIC	Sampling frequency (KHz)	Coupling
	1	Silicone



Cyclic fatigue machine

Transmission Electron Microscope (TEM)

H.L.Huang, 2003



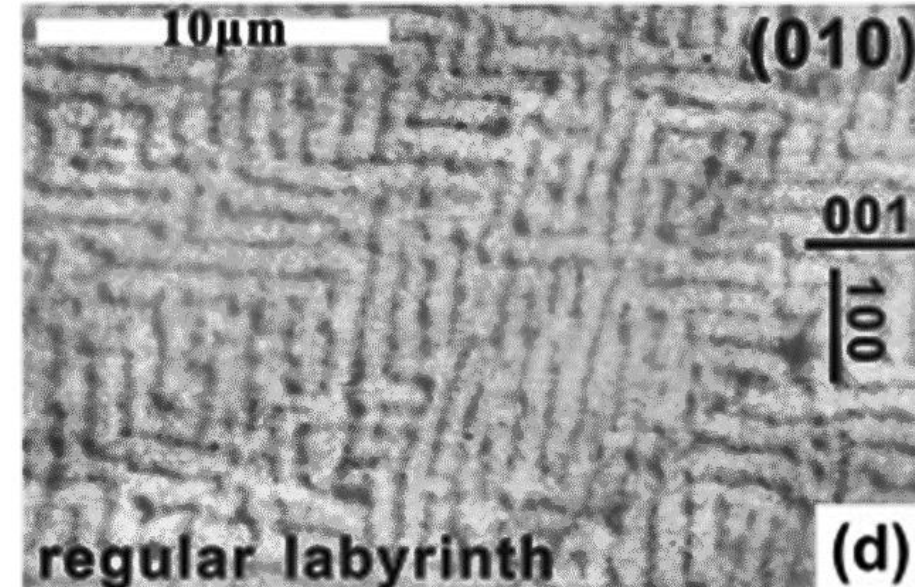
Labyrinth structure in polycrystalline Cu

Thin foil samples

- Tedious preparation
- Good resolution

Scanning Electron Microscope (SEM)

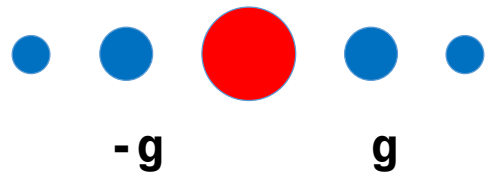
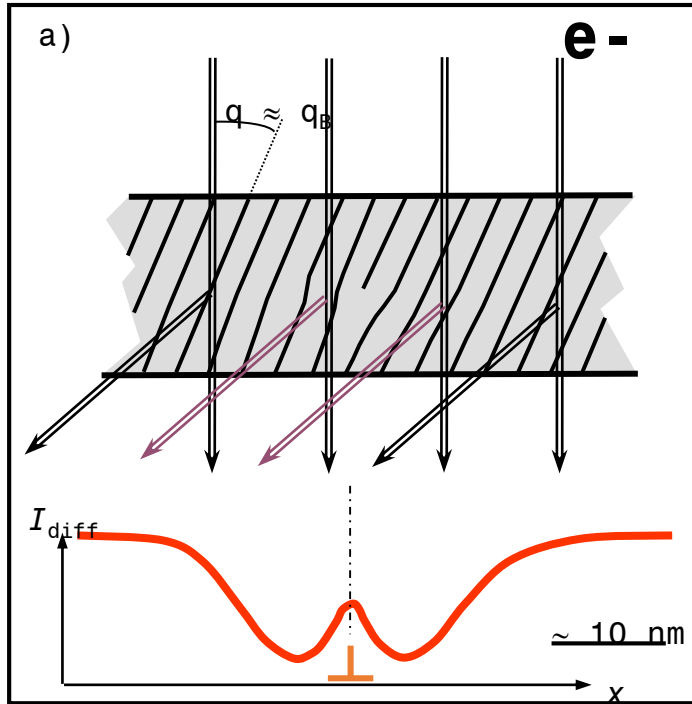
P.Li & al, 2011



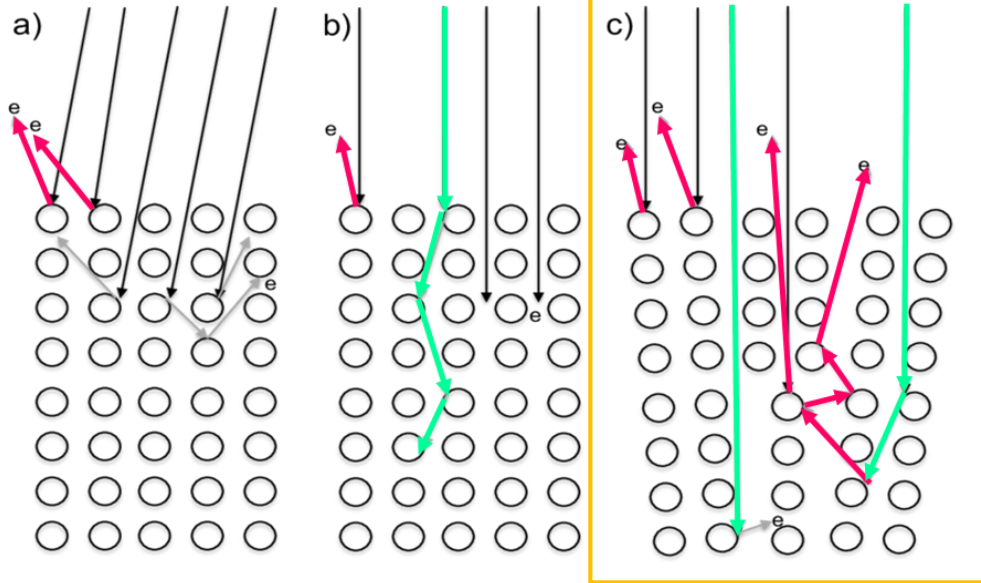
Labyrinth structure in [001] Cu

Applied on bulk materials

- No destructive preparation
- Recent technique

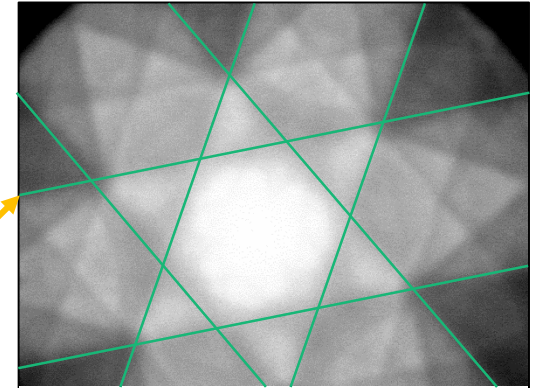


J. Saarimäki, 2015



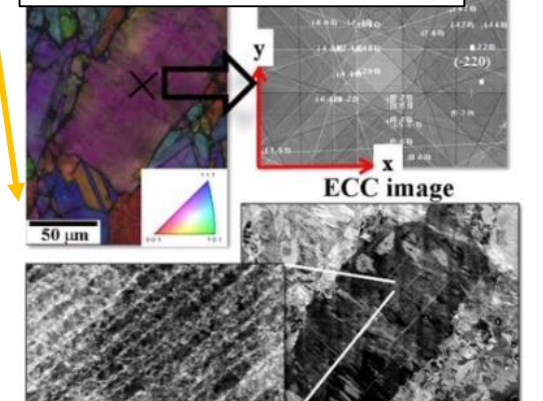
Channeling condition of electrons

- Hard to achieve
 - high precision needed (0.1 – 1°)
- Do not easily allow polycrystal observation



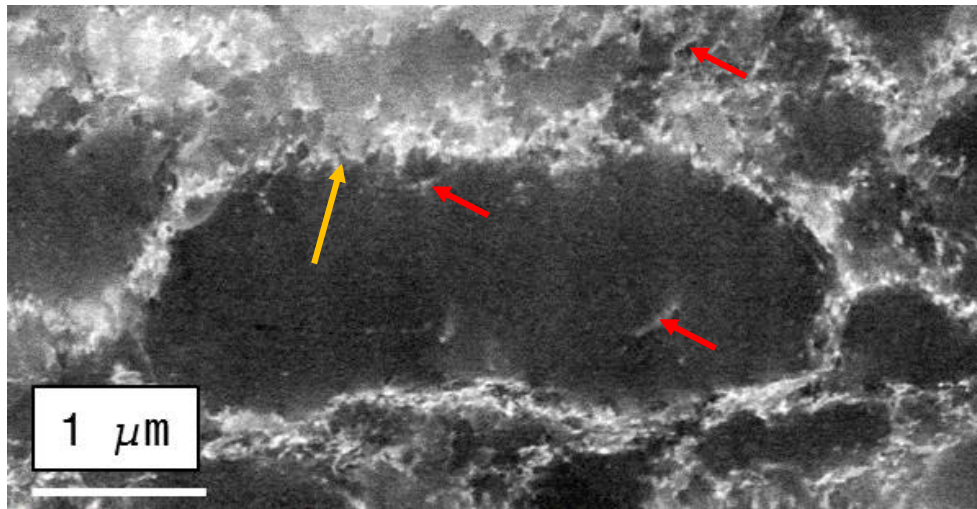
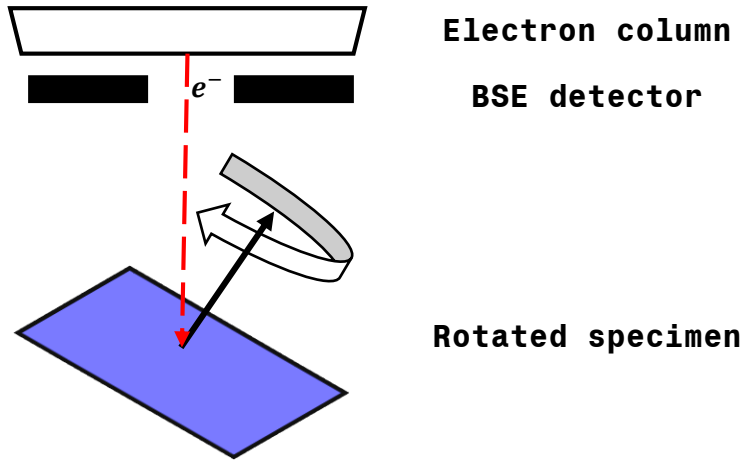
Electron Channeling Pattern

S. Zaefferer, 2014

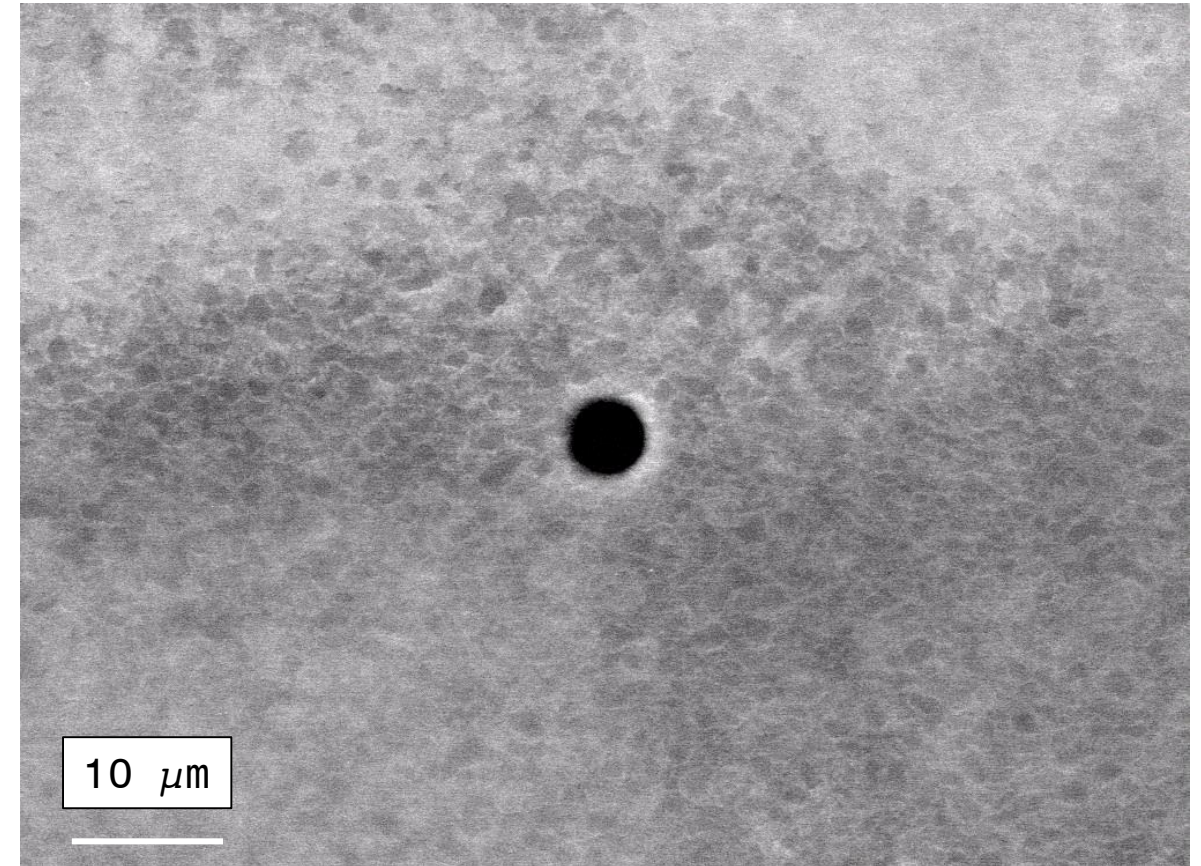


cECCI via EBSD

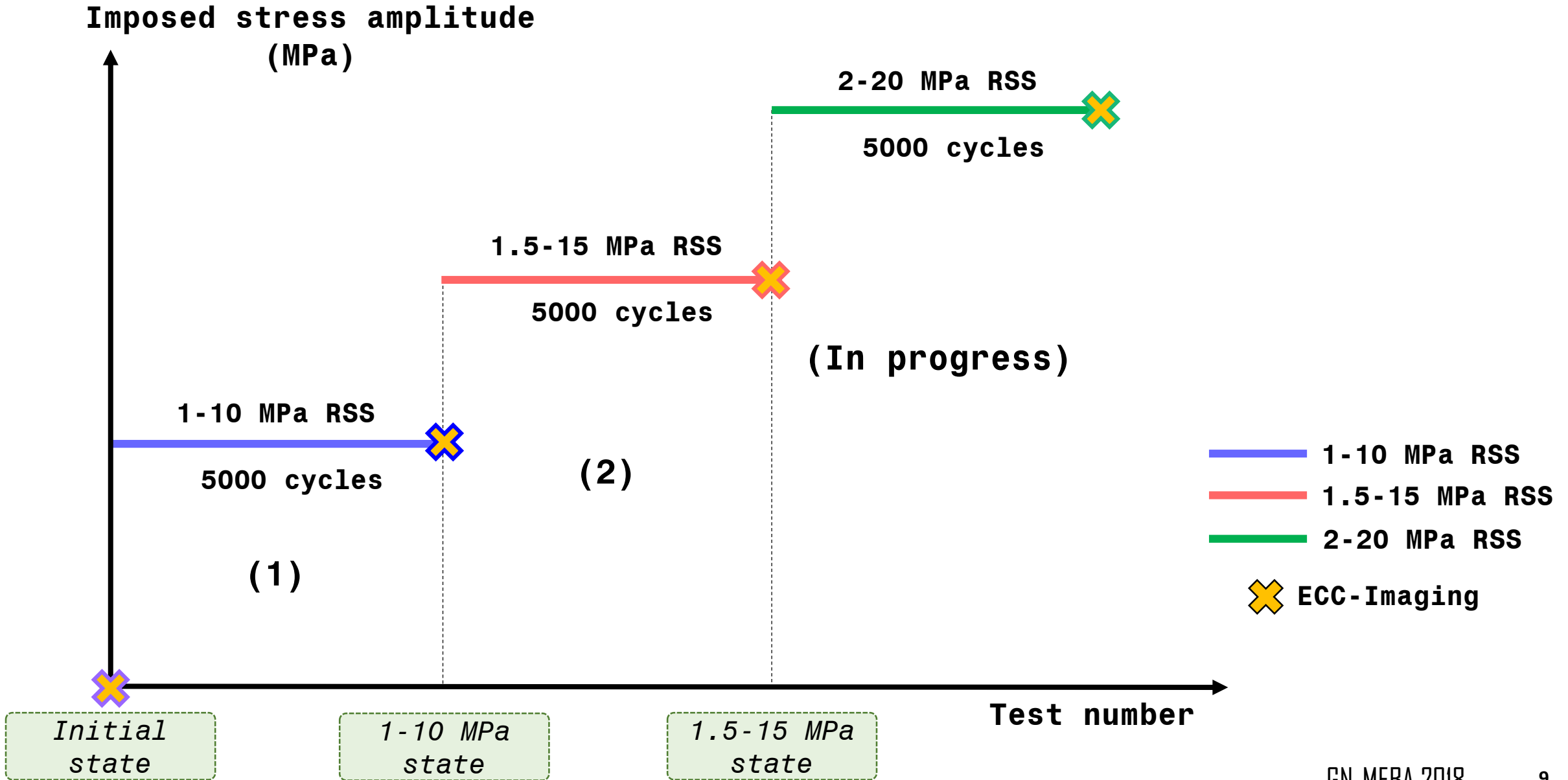
- The idea : Tilt/ rotate & acquire images until dislocations become visible



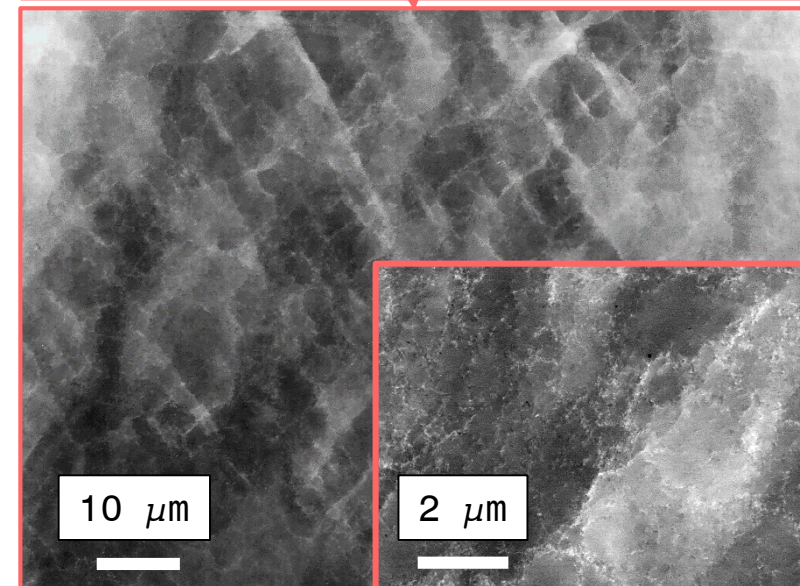
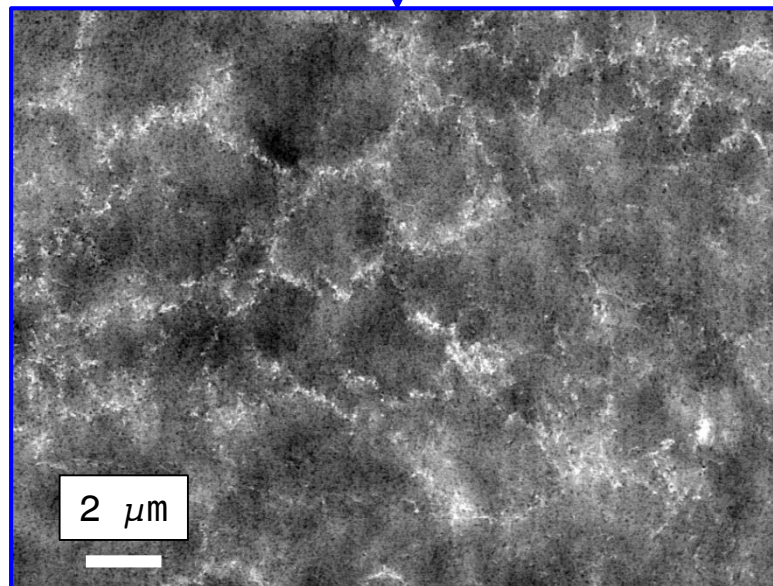
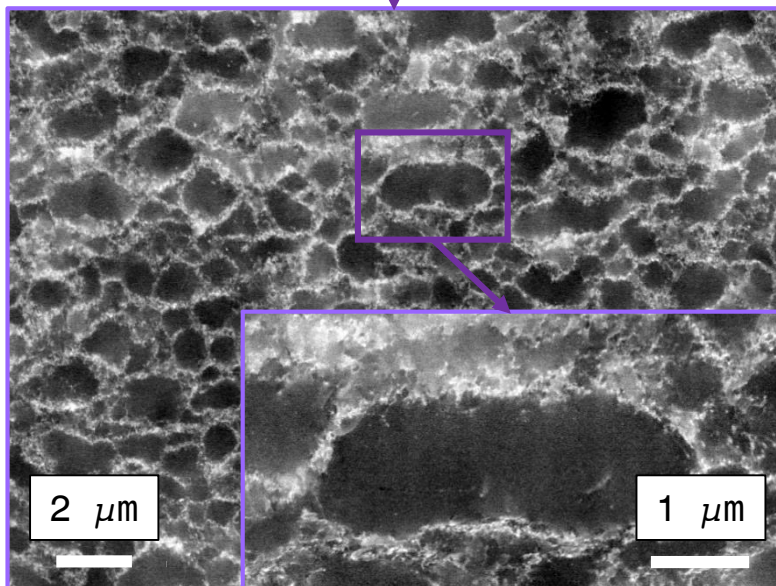
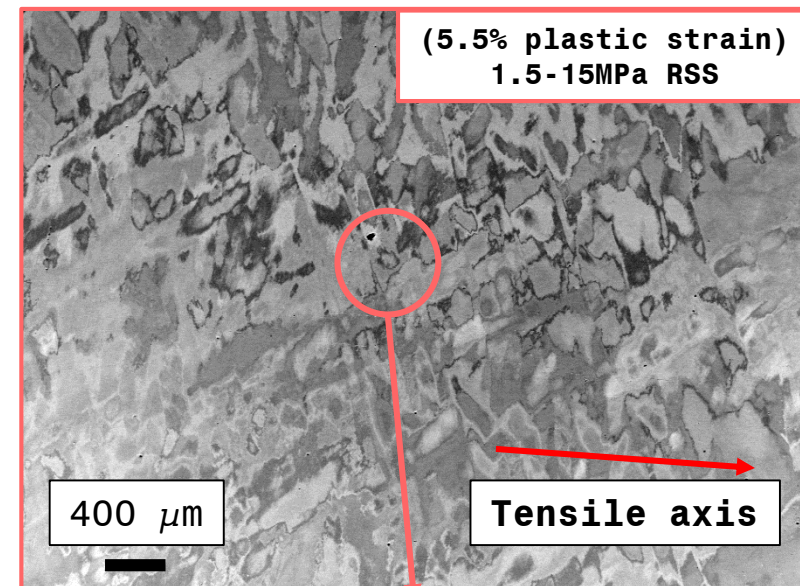
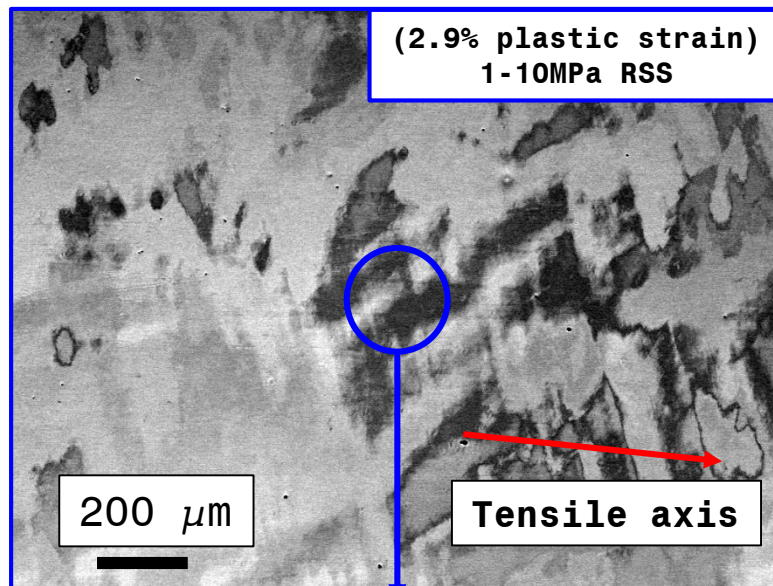
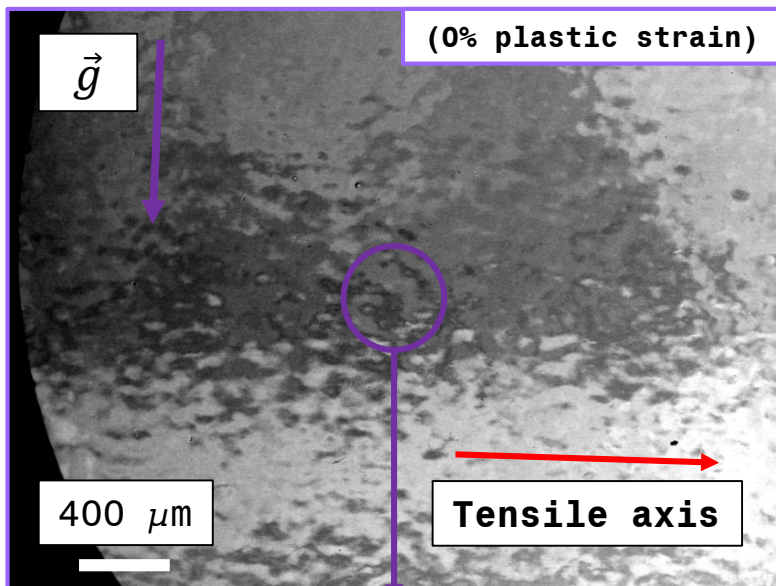
Dislocations cell in Cu single crystal

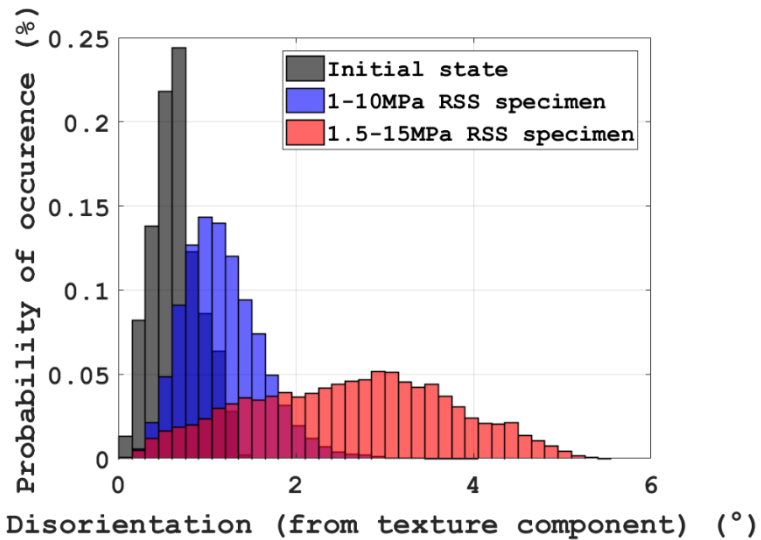
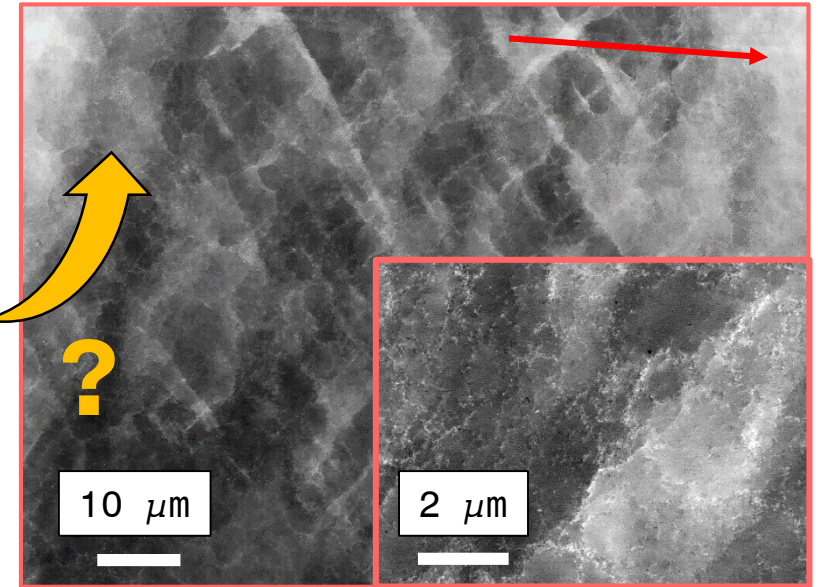
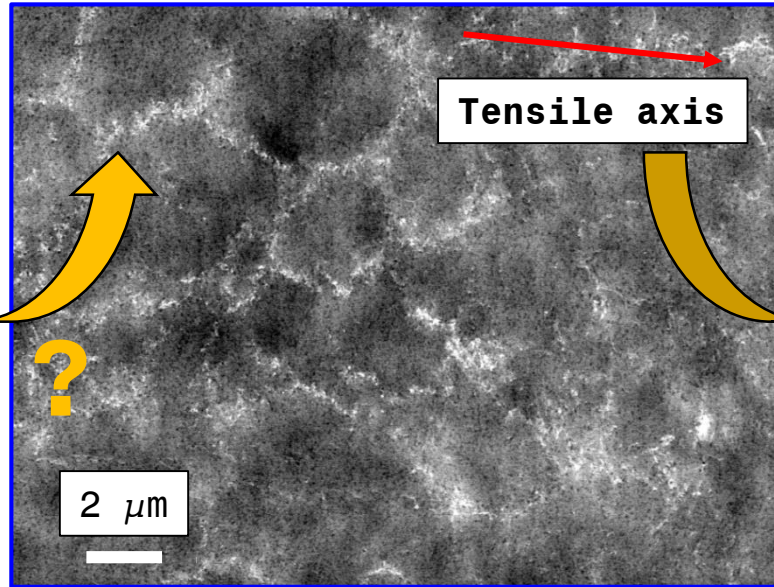
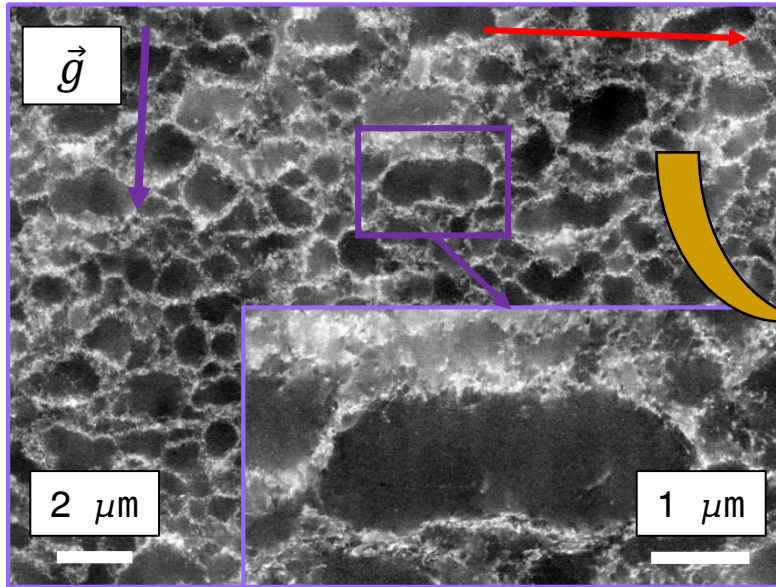


15° of rotation (1° step) – Cu single crystal <110>

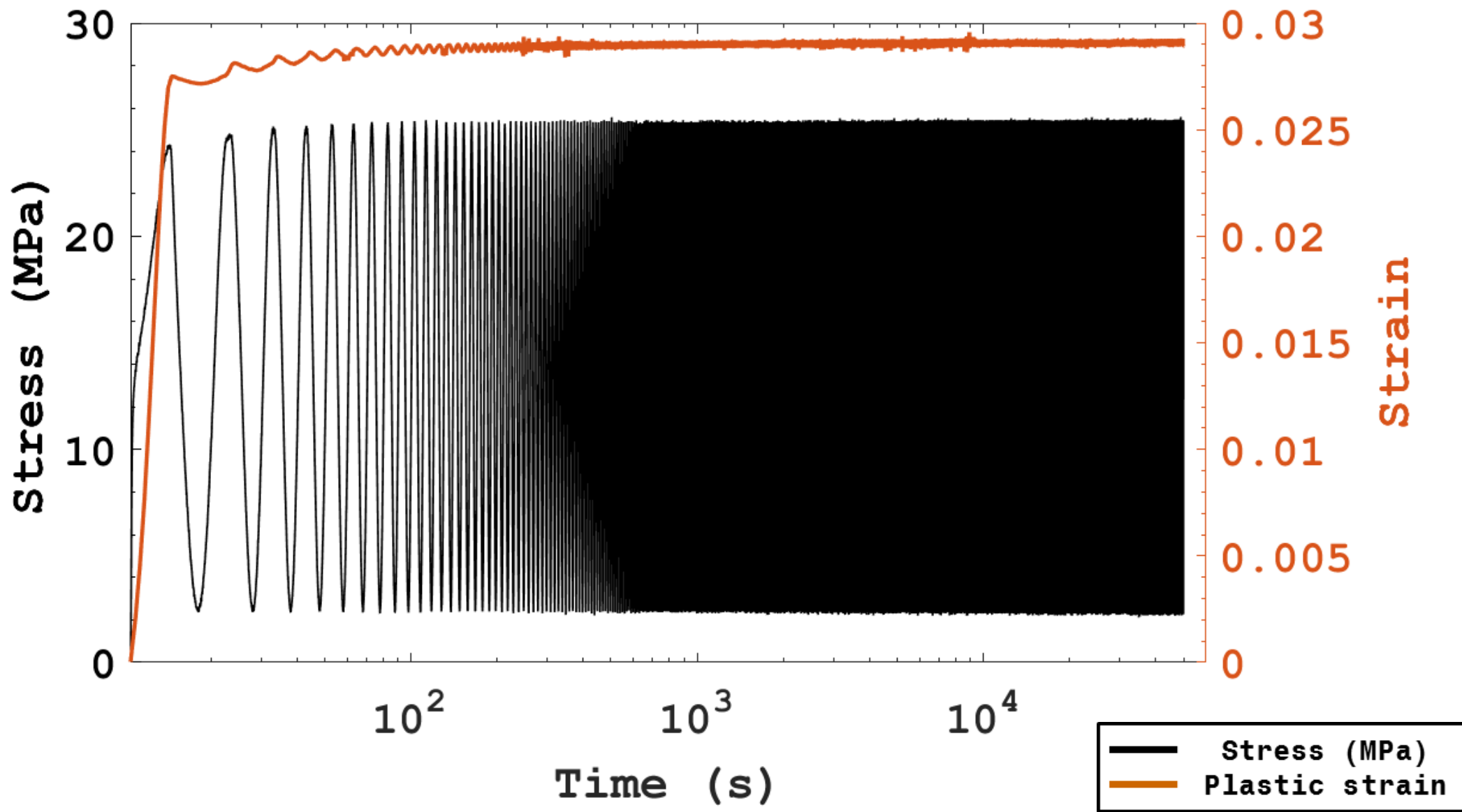


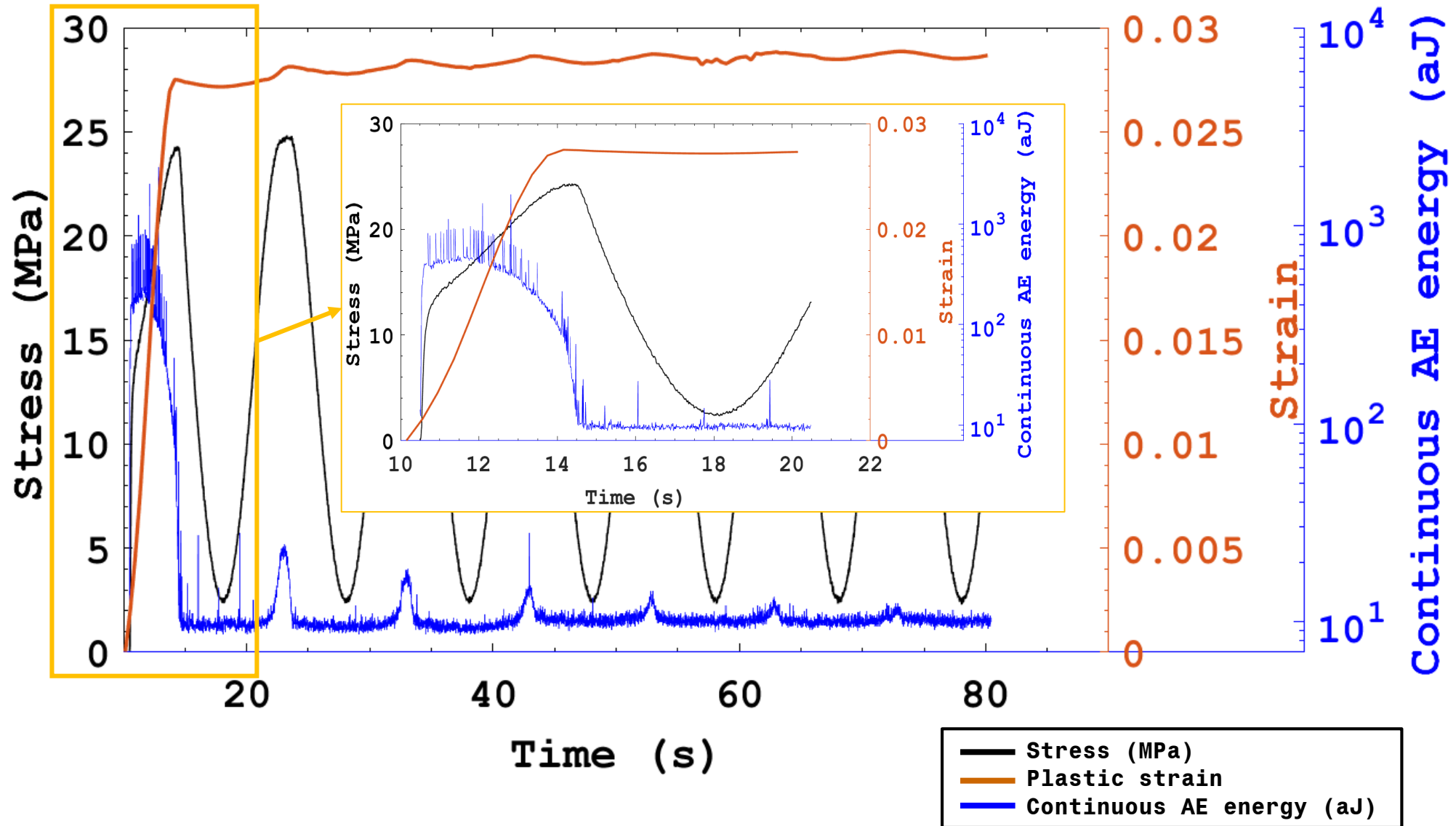
Dislocations structures

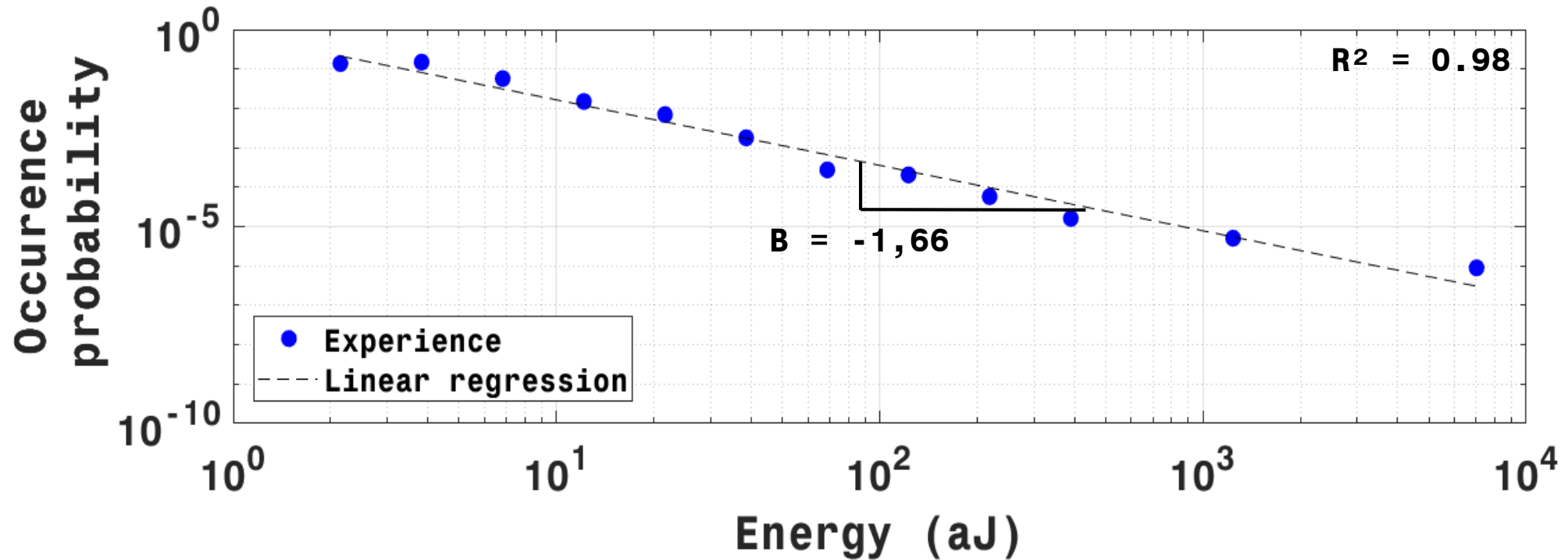




Dislocations structure	Cells Width (μm)	Cells size (μm)	Walls Width (μm)	Walls spacing (μm)
Initial state	$0,16 \pm 0,09$	$0,87 \pm 0,43$	/	/
1-10	$0,36 \pm 0,14$	$1,85 \pm 0,71$	/	/
1.5-15MPa	0.15 ± 0.05	1.95 ± 0.72	$0,51 \pm 0,10$	4.27 ± 1.71



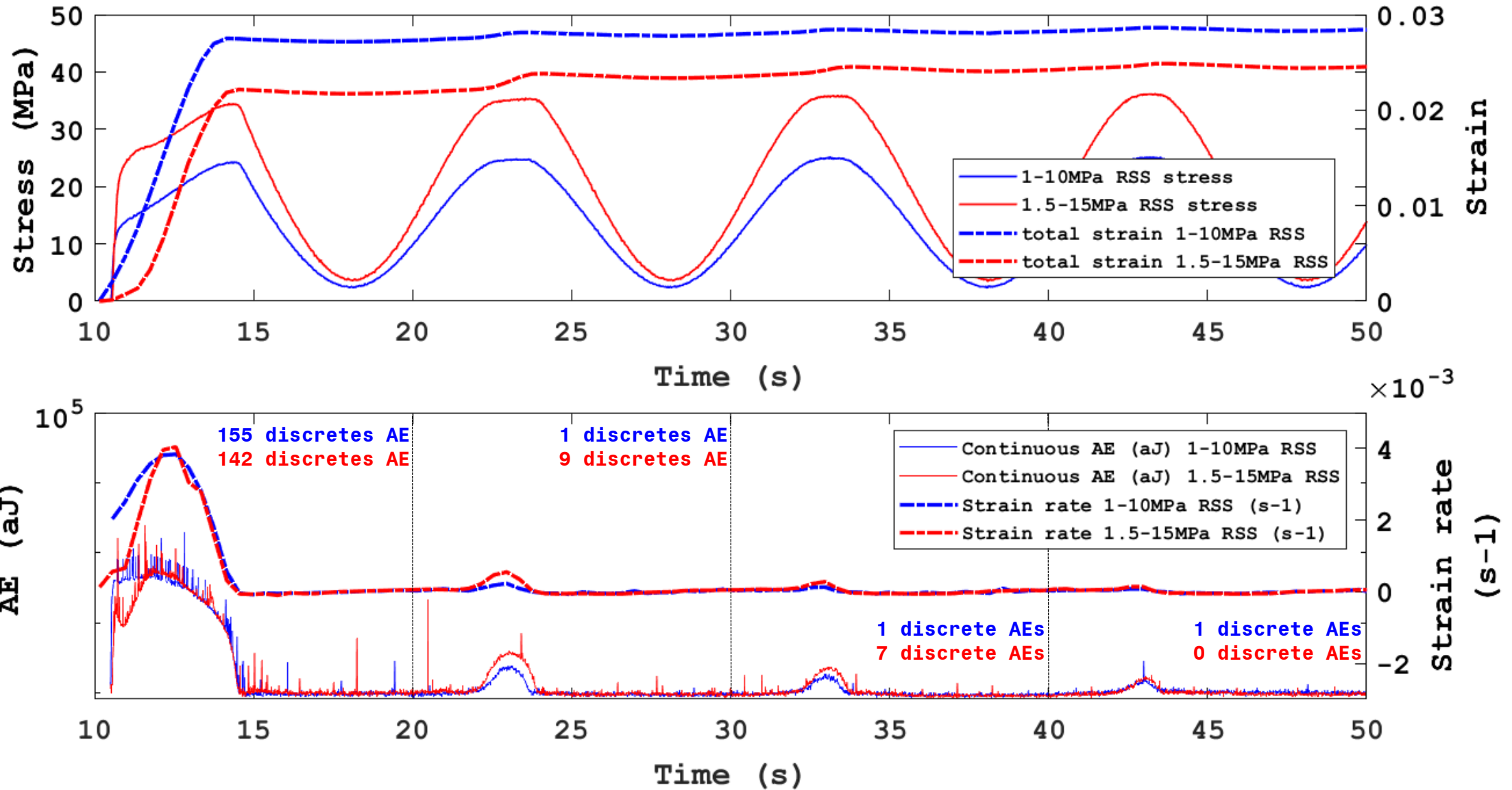


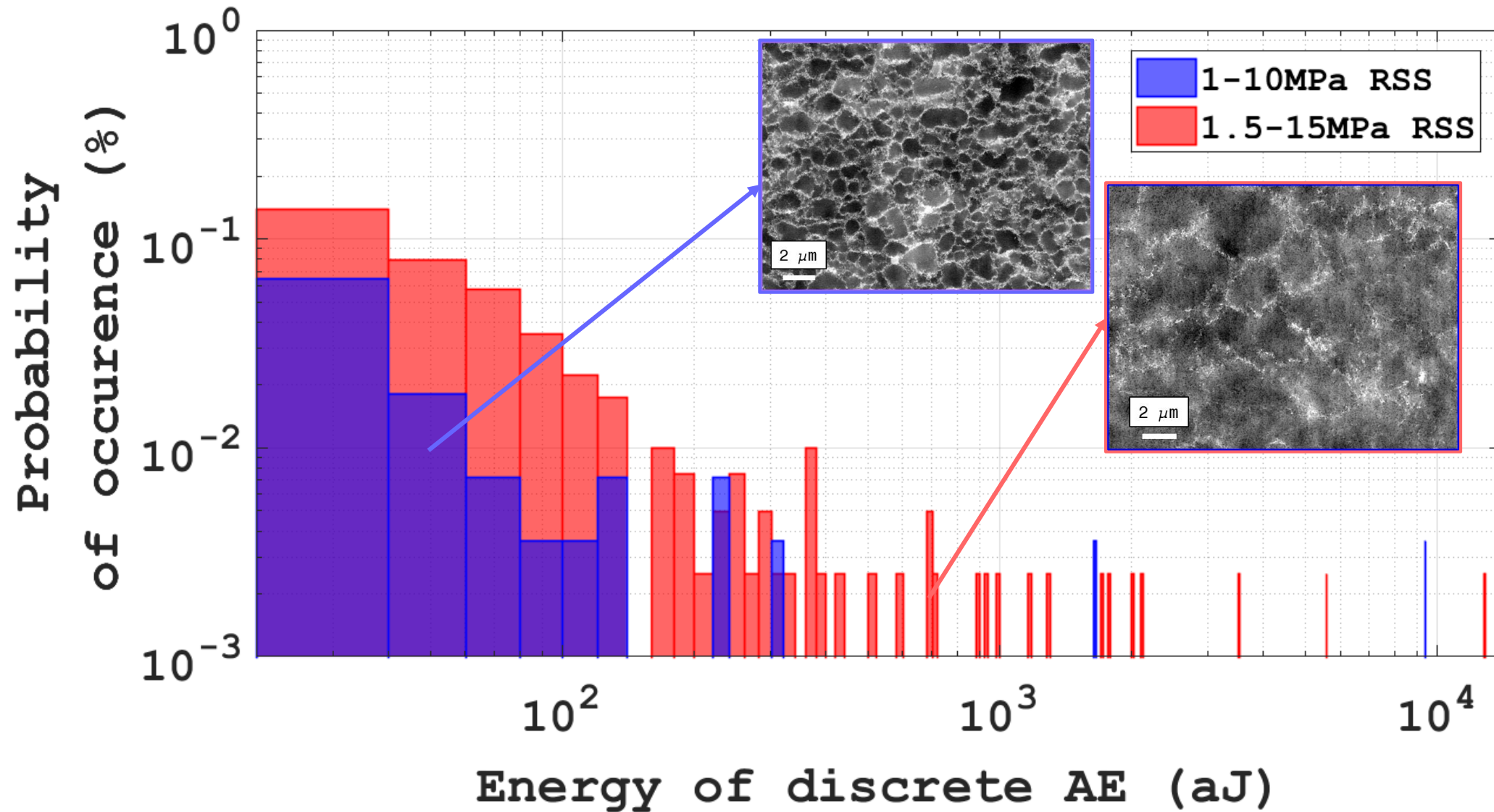


Power-law distribution of discrete AE during the all cycle test

- Power-law distribution
 - exponent in agreement with dislocation avalanches processes (W.Ben.Rhouma,2013)
 - Dislocations avalanches processes

1-10 MPa RSS vs 1,5-15 MPa RSS





- ➔ **Link between AE and dislocation structure is possible.**
 - ➔ **Dislocation avalanches monitored.**
 - ➔ **Initial dislocation structure impact the global dynamics of avalanches.**
- ➔ **« Rotation ECCI » : An easy way for observing dislocations.**

- ➔ **Stress imposed fatigue test.**
 - ➔ **Evolution of the structure and the AE ?**
- ➔ **Influence of crystallographic orientation**
 - ➔ **Multiple slip / Single slip**
- ➔ **Improve the « Rotation ECCI » apparatus**
 - ➔ **Determine the diffraction vector \vec{g}**

Thanks for your attention