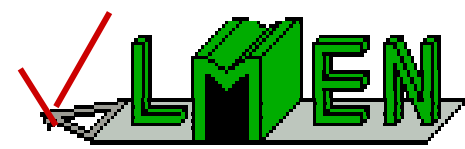




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Champagne-Ardenne

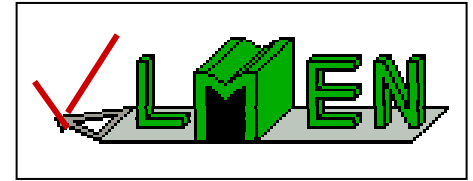


## ***Les potentialités de la nano-EBIC***

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*LMEN, Laboratoire de Microscopies et d'Etude de Nanostructures,  
EA 3799, Université de Reims Champagne Ardenne,  
21, rue Clément Ader, 51685 Reims Cedex 2, France.*





# Outline

- **Introduction**
- **Principes physiques de l'EBIC**
- **Qu'est-ce que la nano-EBIC ou EBIC en champ proche ?**
- **Instrumentation: combinaison AFM/MEB**
- **Exemples d'applications**
- **Conclusion**

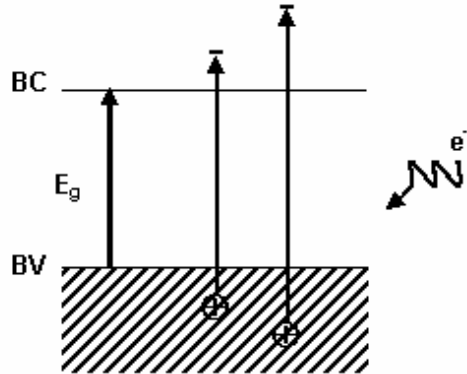
# Introduction

## EBIC: electron beam induced current

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- Electron beam induced current (EBIC) method is a powerful characterization tool for the analysis of electronics devices
  - used for imaging the electrically active defects and heterogeneities in semiconductors
  - used for determining the diffusion length of minority carriers
  - used for estimating recombination velocities at grain boundaries and surfaces
- Resolution of EBIC imaging is chiefly limited by the lateral dimension of the energy dissipation volume, which depends on the material and primary electron beam energy
- Since the minimum feature sizes of electronic devices are continuously decreasing to enhance packing density and speed, and reduce power consumption there is a need for improved resolution
- That is the reason why we have developed a **new technique that we call near-field EBIC or nano-EBIC**

# Principes physiques de l'EBIC



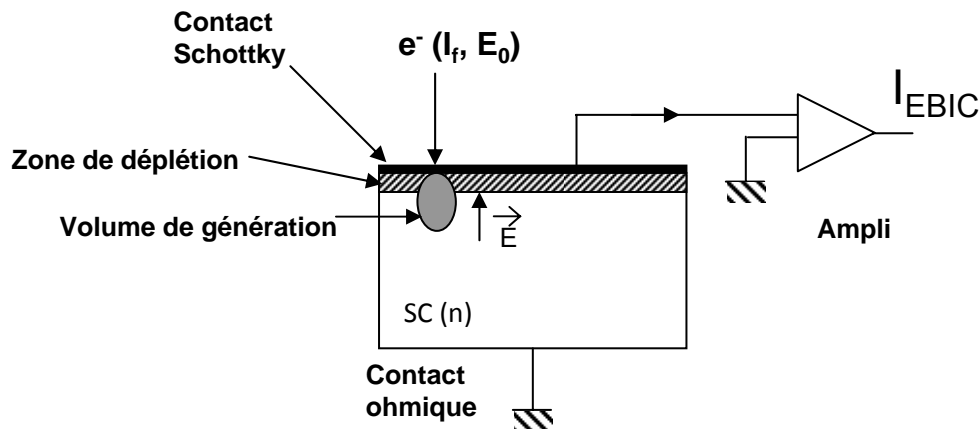
Mécanisme de création des paires électron-trou.

Processus possible si  $E_0 > E_G$

Nombre de paires défini en terme de rendement quantique

$$Q = E_0 / E_i$$

$E_i$  = Energie d'ionisation  $\cong 3 E_G$

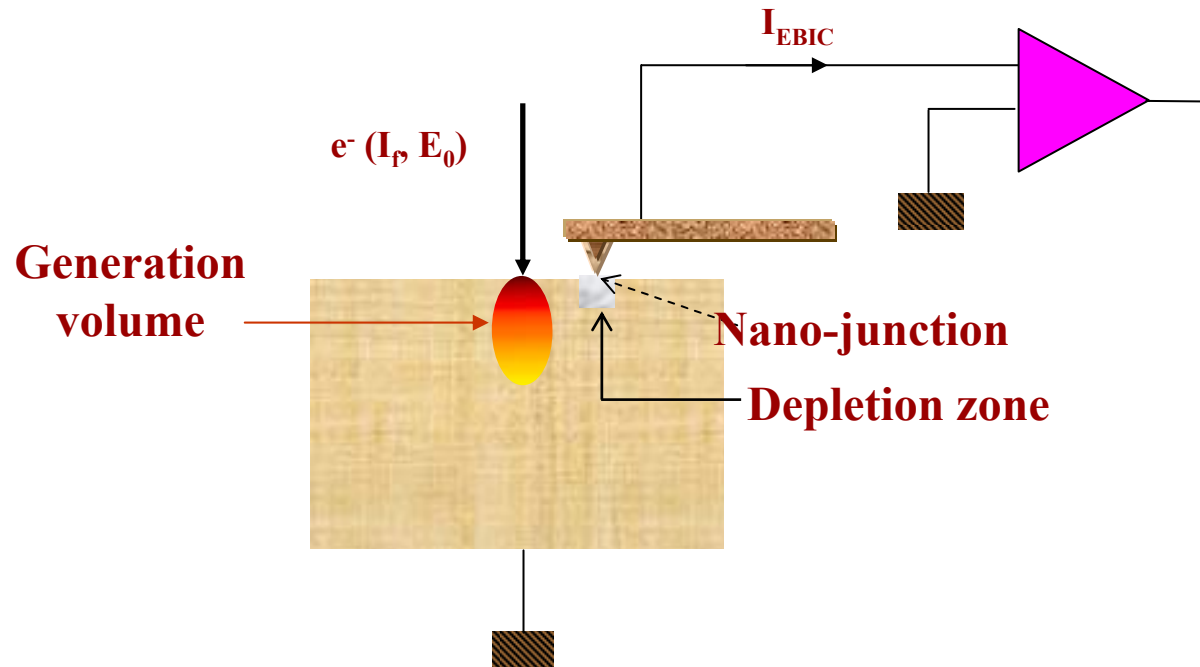


En absence de champ électrique diffusion et recombinaison des porteurs

En présence de champ, l'excédent de porteurs minoritaires va se propager et former le courant  $I_{EBIC}$

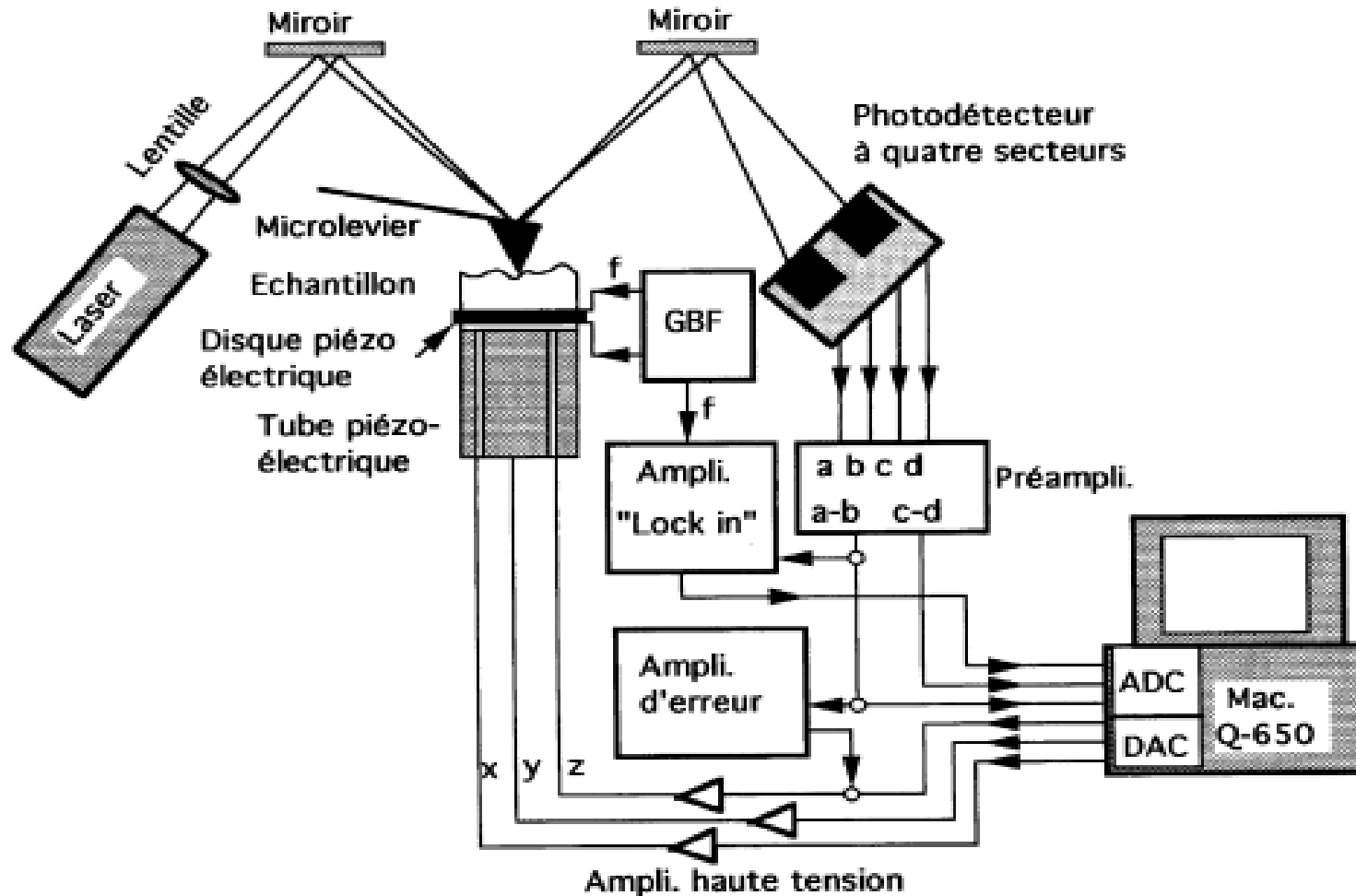
Principe Général de l'EBIC dans le cas d'une diode Schottky.

# Principle of nano-EBIC

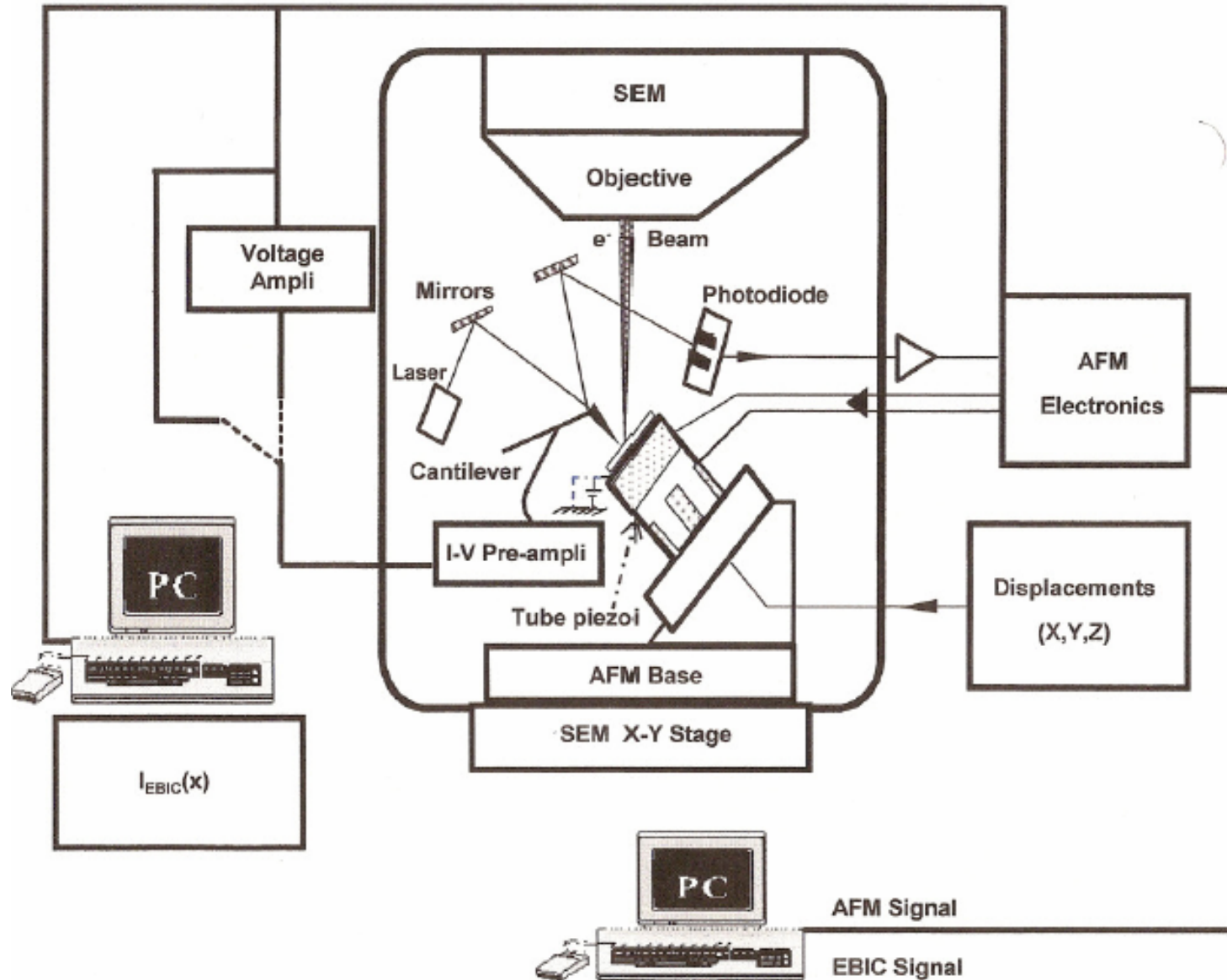


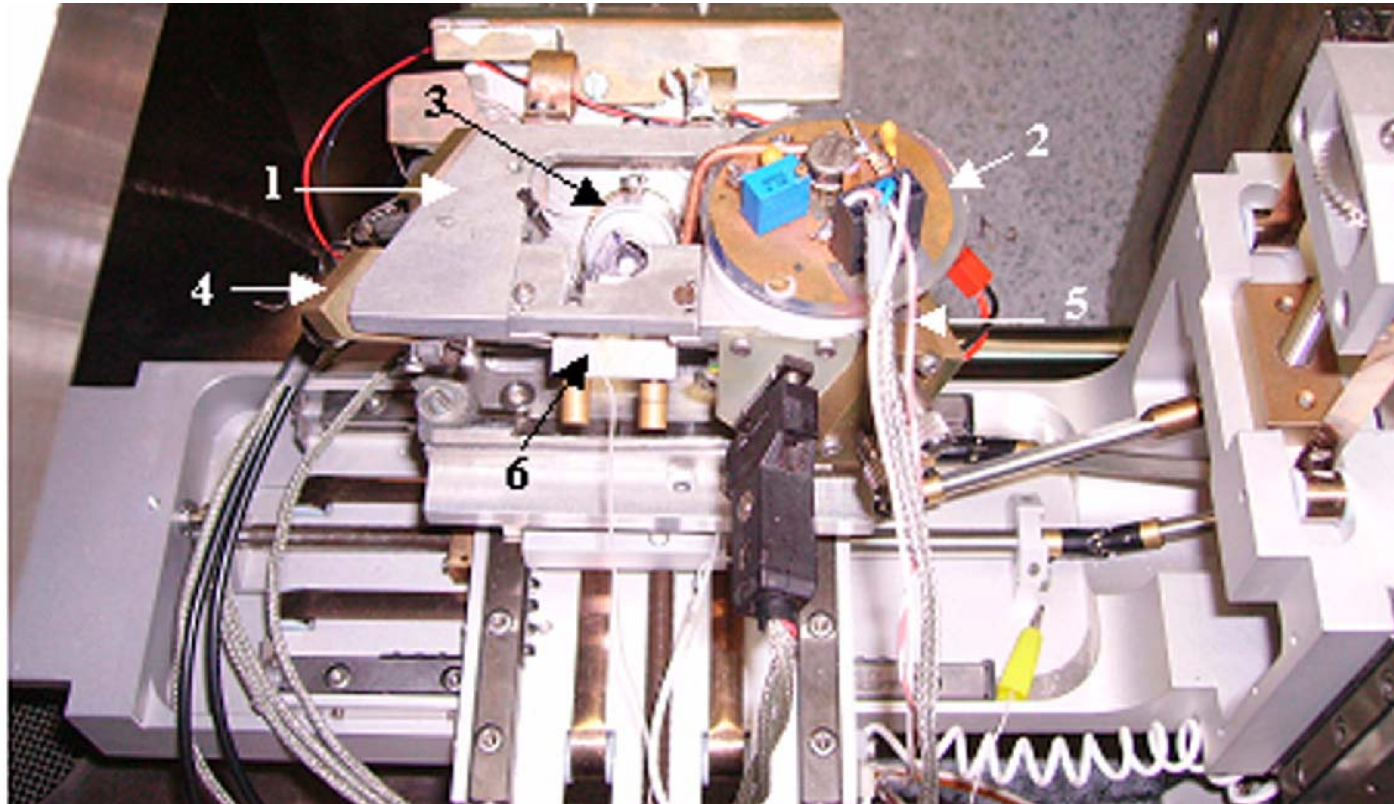
- ➡ **Resolution is not limited by the generation volume**
- ➡ **Resolution is limited by the dimension of the depletion zone ~ 10 nm**

# Schematics of the AFM



# The Combined AFM/SEM system





**View of the AFM mounted inside  
the specimen chamber of the SEM**

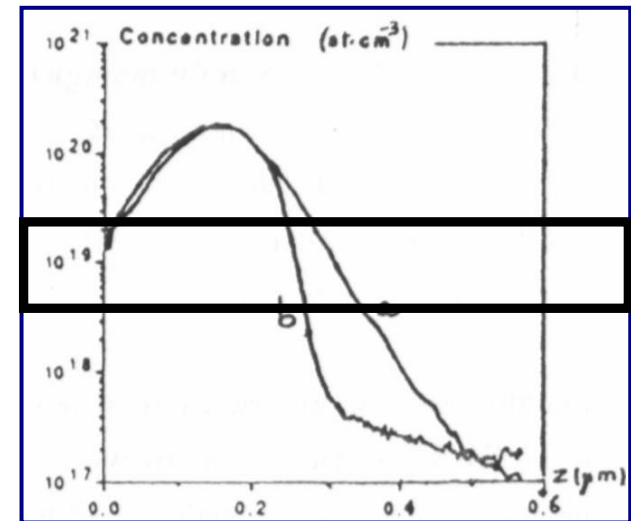
# Application to shallow p<sup>+</sup>-n junction

K. Smaali *et al.*, *Ultramicroscopy* 108 (2008) 605.

## Description of the p<sup>+</sup>-n junction

- ▶ Preamorphisation of Si (n) substrate on 240 nm depth (Si<sup>+</sup>, 150 keV, 5.10<sup>15</sup> ion.cm<sup>-2</sup>)
- ▶ The p-doped layer is obtained with B<sup>+</sup> implantation (at 50 keV, 5. 10<sup>15</sup> ion.cm<sup>-2</sup>)
- ▶ Rapid thermal annealing (< 1s, 1000 °C)

**SIMS profile of p-doped layer**  
(50 keV, 5.10<sup>15</sup> ions.cm<sup>-2</sup>)

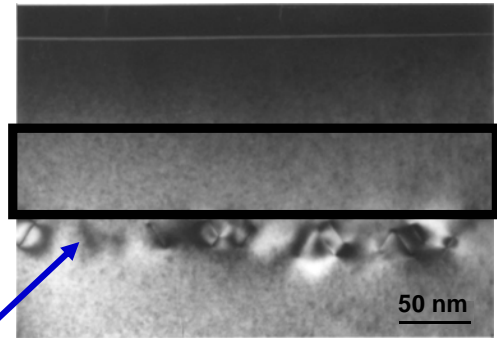


(a) substrate without preamorphisation  
(b) substrate after preamorphisation

## Advantages

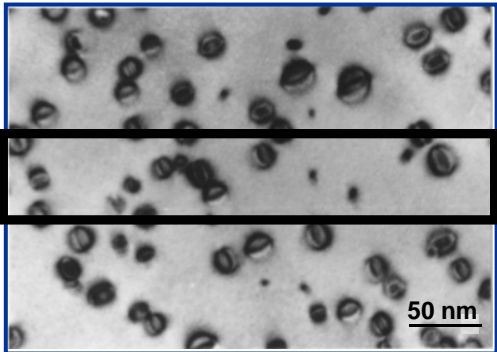
- ▶ Elimination of extended defects
- ▶ Avoid B<sup>+</sup> diffusion during activation step
- ▶ Reduce the doping profile to about 250 nm

Surface 0  
Si (p<sup>+</sup>)  
240  
Z (nm)



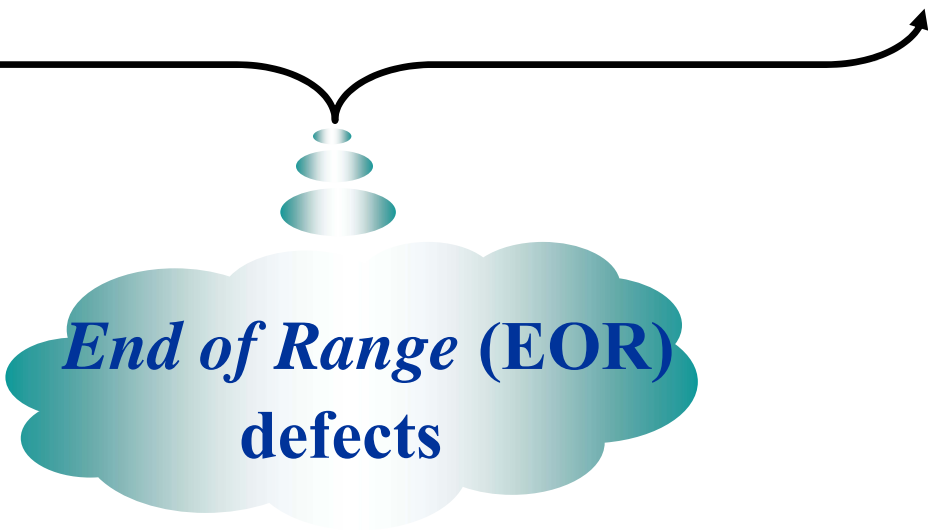
Dark-field TEM Image  
Cross-section

- Dislocation loops :  
**Interstitial type**
- **Non homogeneous  
Distribution**



TEM Image  
Plane-view

End of  
range  
defects

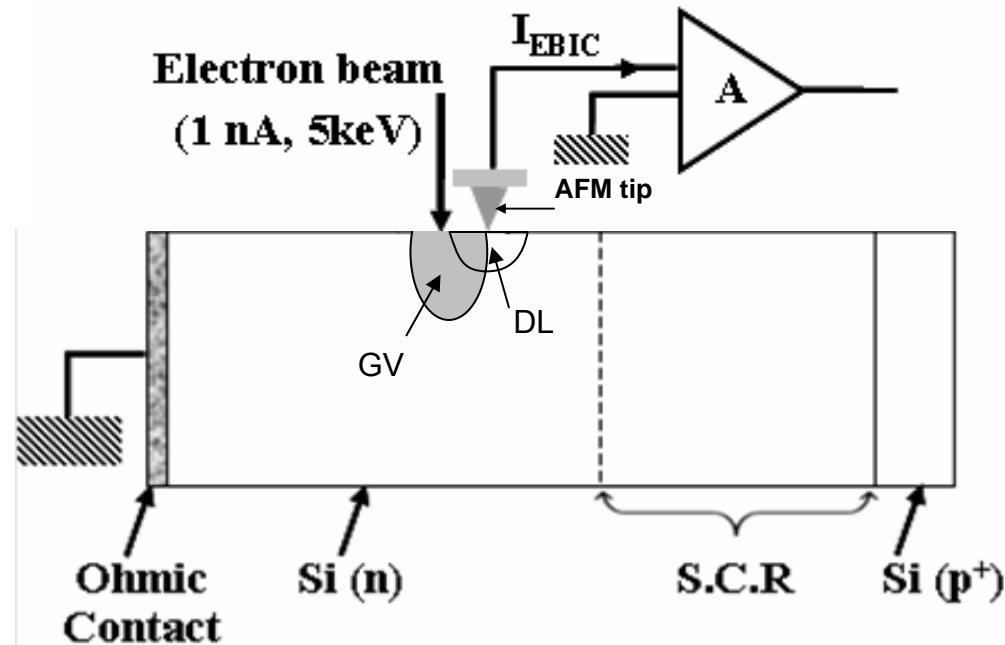


Disadvantage

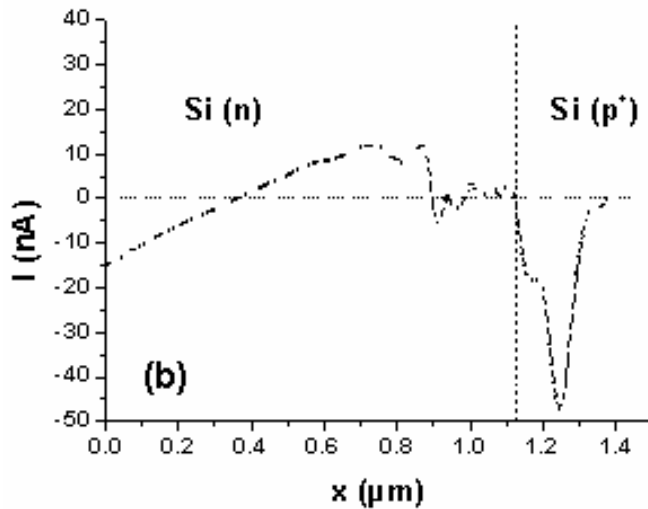
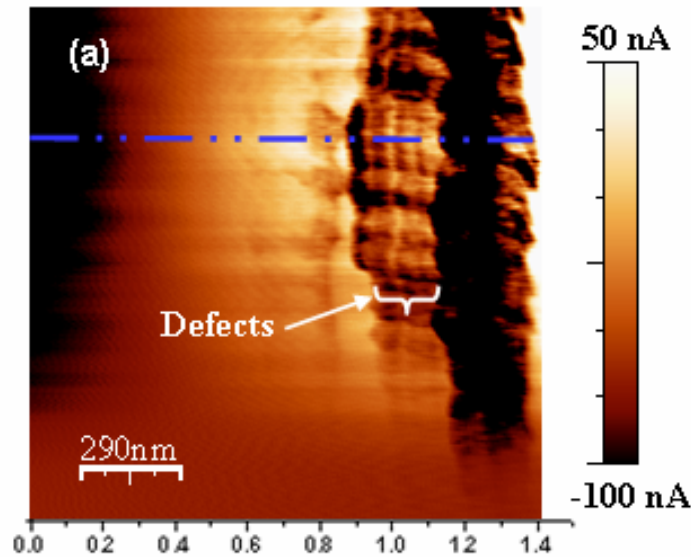
**➡ Formation of a band of defects**

# Study of the p<sup>+</sup>-n junction in cross-section

## Set-up



- Existence of a small hemispherical Depletion layer (DL) at tip-surface contact which constitutes a MOS junction due to surface native oxide
- Space Charge Region (SCR) due to p-n junction
- When the tip is positioned above the SCR the carriers are subjected to 2 electric fields in opposite direction



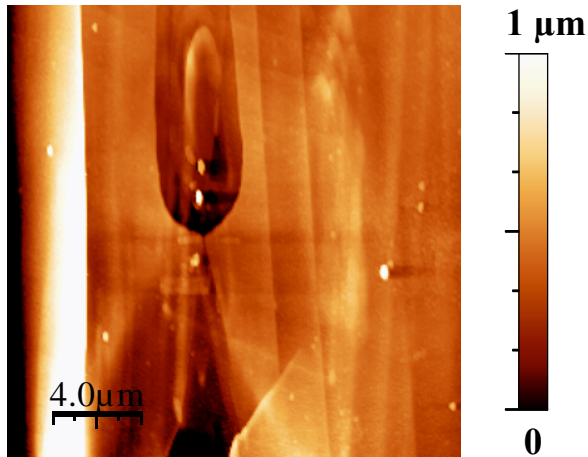
- This example illustrates also the high-resolution of nano-EBIC
- Contrast variations are detected at the interface between n and p-doped regions, probably due to electrically active defects that are the interstitial dislocation loops revealed by TEM

- Resolution of about 20 nm

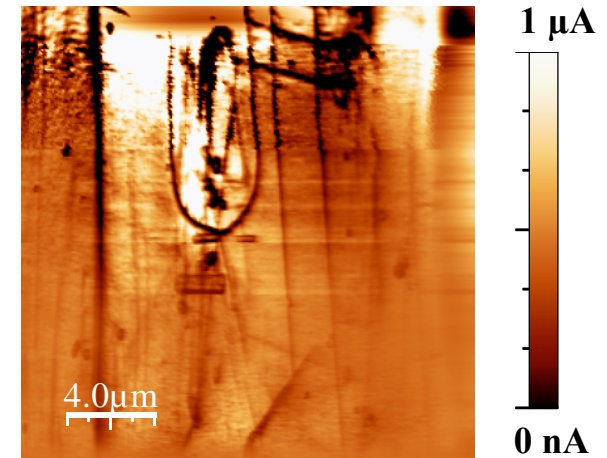
# Application to III-V Heterostructures

K. Smaali and M. Troyon, *Nanotechnology*, 19 (2008) 155706

## GaAs sample Indented with Berkovich diamond tip



Topography



EBIC image  
 $I_f = 1 \text{ nA}$ ,  $E = 5 \text{ keV}$

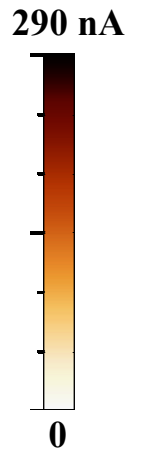
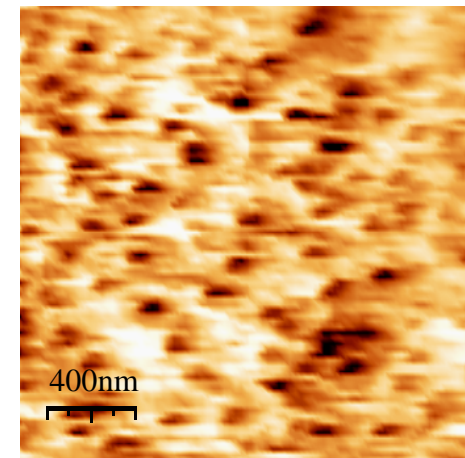
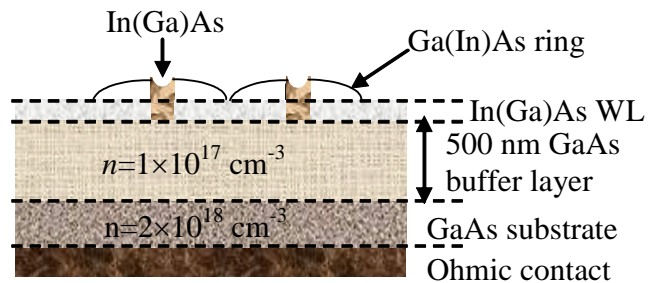
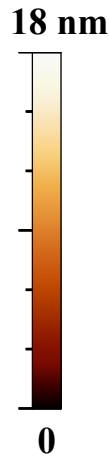
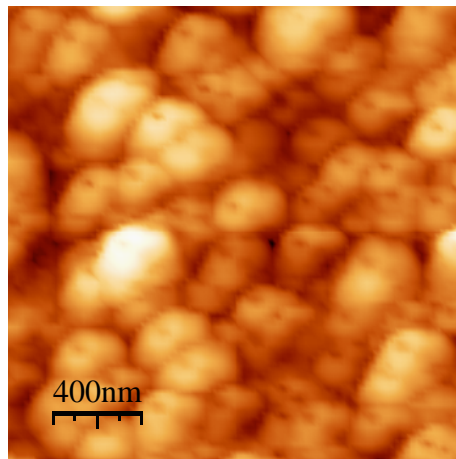
Dislocations or slip bands are clearly resolved in the EBIC image

Carriers are trapped on the dislocations due to non radiative recombination

# InAs/GaAs quantum dots

M. Troyon and K. Smaali, Nanotechnology, 19 (2008) 255709.

The QDs are grown by MBE and capped with a 5 nm thick GaAs layer



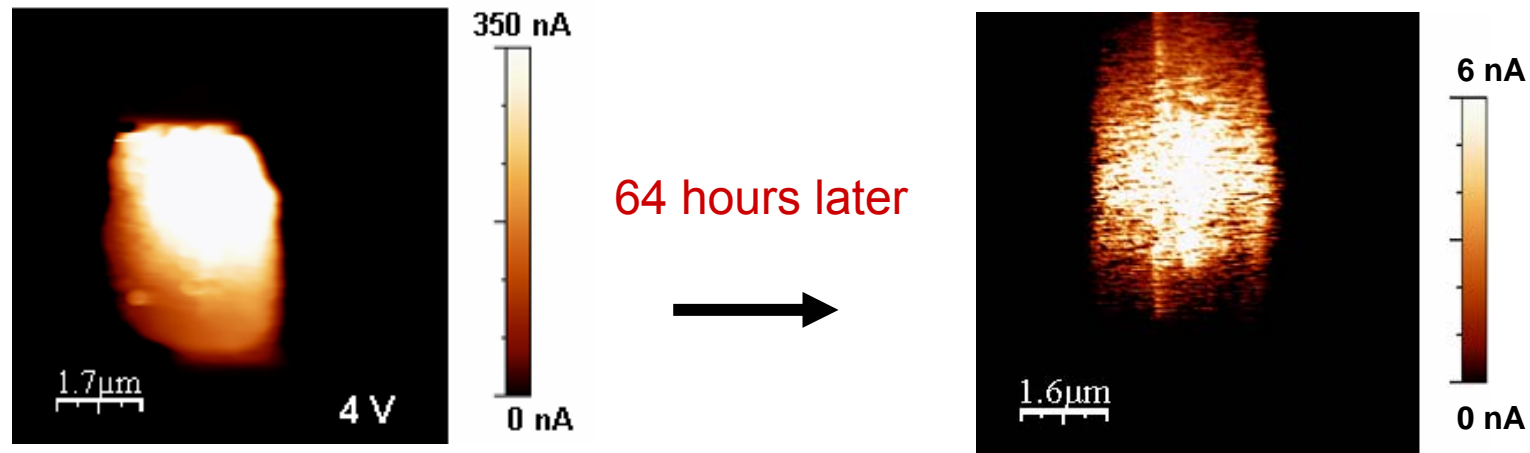
Topography

EBIC Image  
 $I_f = 1 \text{ nA}$ ,  $E = 5 \text{ keV}$

The cap-layer forms a ring-like structure around the QD

The EBIC image shows that the current is very high in the middle of this nano-structure, which is the QD

# Writing-erasing-memory effects



C-AFM electric image of InAs/GaAs QD sample after electron irradiation of the central part at  $I = 1$  nA and  $E = 5$  keV.

## WRITING

Holes are trapped in QDs and surface states during acquisition of the EBIC image (central part). Thanks to holes trapping, unpinning of the Fermi level occurs and then a C-AFM image can be obtained

## ERASING

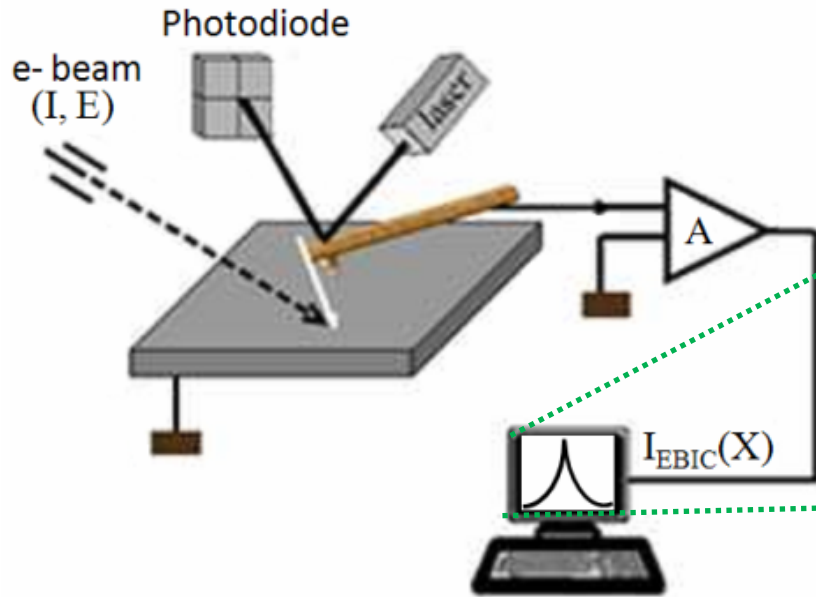
Erasing occurs when electron irradiation current is weak ( $I=60$  pA). In these conditions surface recombination velocity is high and no induced current is generated. There is untrapping of the holes and thus Fermi level pinning, which hinders C-AFM image to be obtained even at  $V=10$  V.

## MEMORY

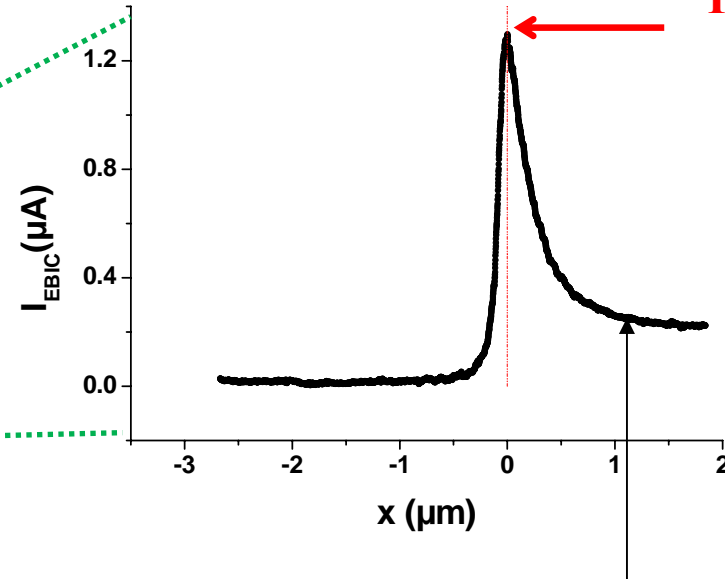
A strong current is still obtained after 64 hours waiting, evidencing a very efficient memory effect. The large charge retention time is explained by the presence of a thick depletion layer at the surface sample preventing the charges trapped in the QDs and the surface states from flowing towards the substrate.

# Measurement of diffusion length

## Experimental procedure



EBIC profile for GaAs  
( $I_f = 1$  nA,  $E = 5$  keV)



Position of the tip

$I_{EBIC} \propto \exp(-x/L_{eff}) \longrightarrow L_{eff} = \text{effective Diffusion length}$

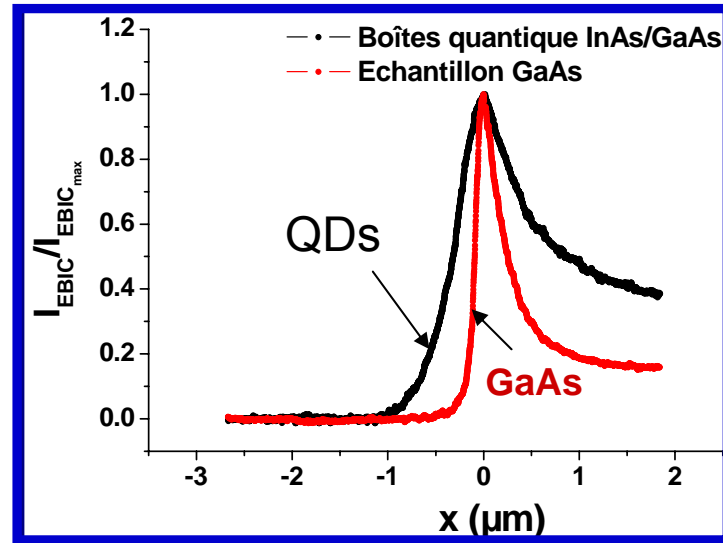
$$(L_{eff})_z^2 = L^2 \left\{ 1 - \left[ \frac{s}{1+s} \right] \exp(-z/L) \right\}$$

$(L_{eff})_z$  is the effective diffusion length of the small volume at depth  $z$

$L$  is the bulk diffusion length and  $s = S\tau/L$  with  $S$  the surface recombination velocity and  $\tau$  the carrier lifetime

Photon recycling due to photon reabsorption caused by radiative recombination

# Influence of the presence of QDs



GaAs reference sample  $\rightleftharpoons$   $L=0.12\mu\text{m}$       InP reference sample  $\rightleftharpoons$   $L=0.26\mu\text{m}$   
GaAs containing QDs  $\rightleftharpoons$   $L=0.40\mu\text{m}$       InP containing QDs  $\rightleftharpoons$   $L=0.46\mu\text{m}$

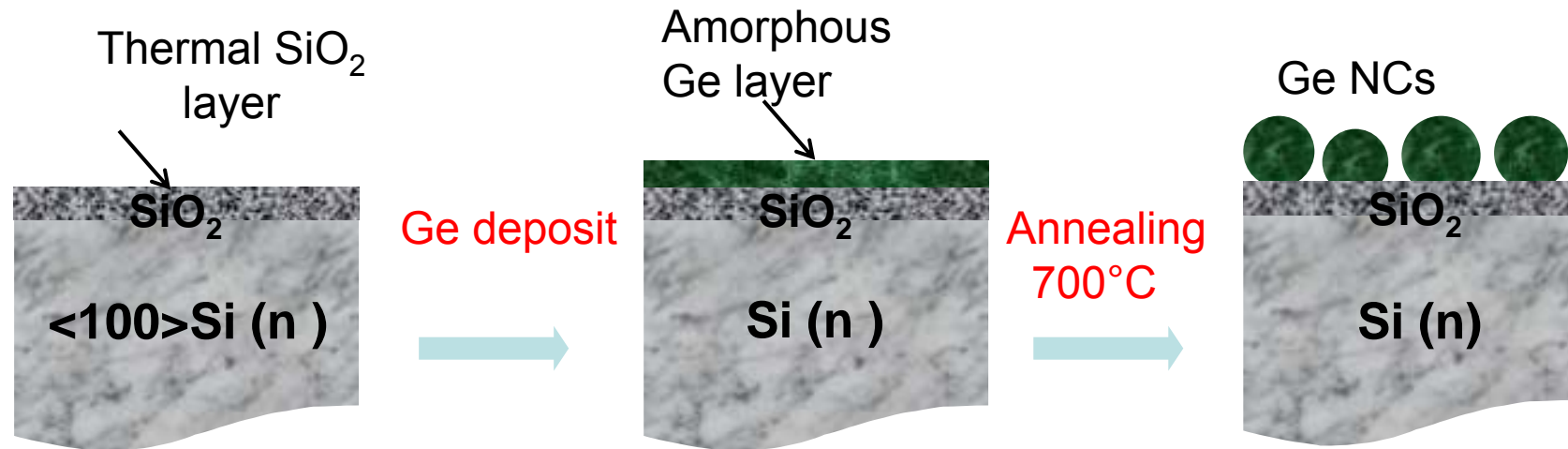


**Lateral exchange of carriers via the wetting layer  
because QDs and surface states are saturated**

# Germanium nanocrystals grown on SiO<sub>2</sub>

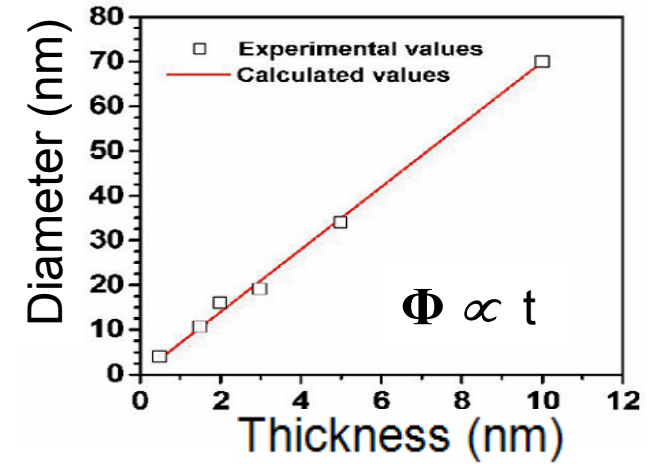
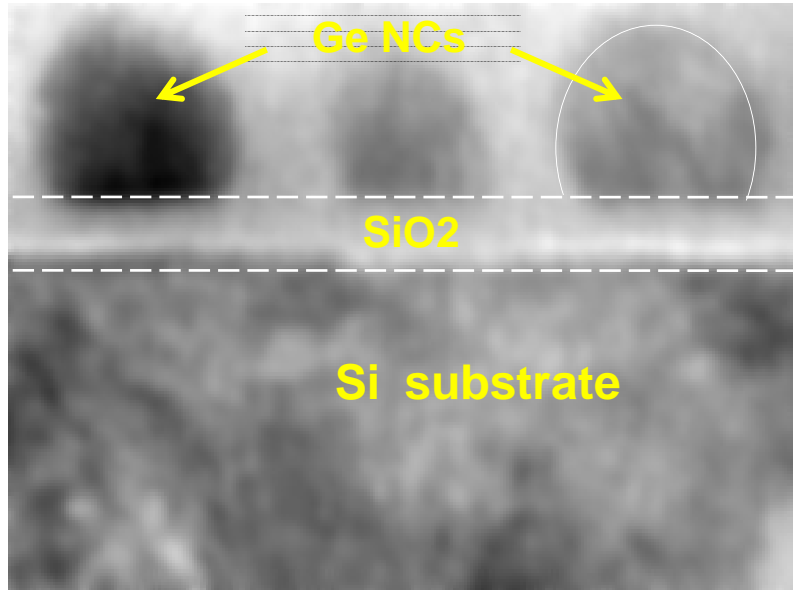
P.D.Szcutnik *et al*, Eur. Phys. J. Appl. Phys. 41 (2008) 103

## Sample preparation



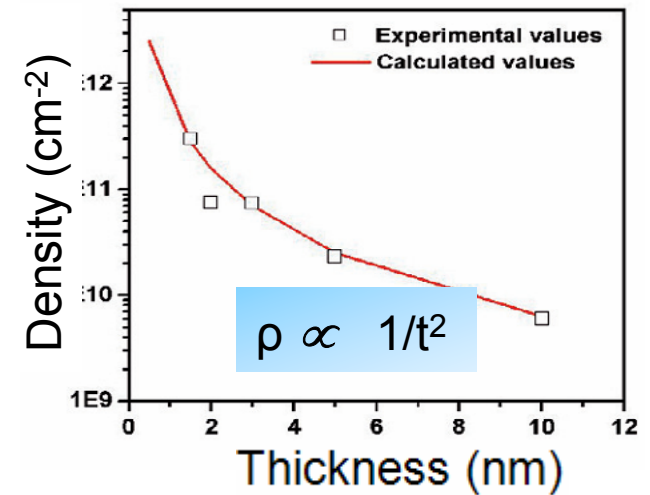
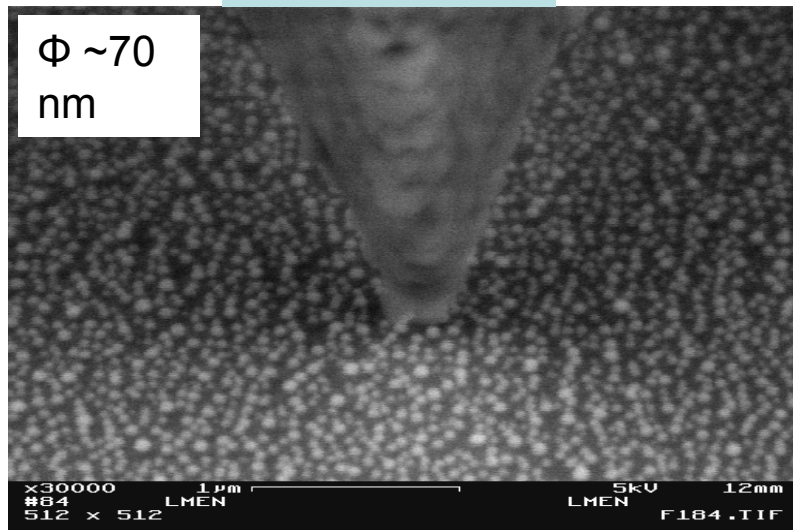
MBE chamber (10<sup>-11</sup> torr)

## TEM image



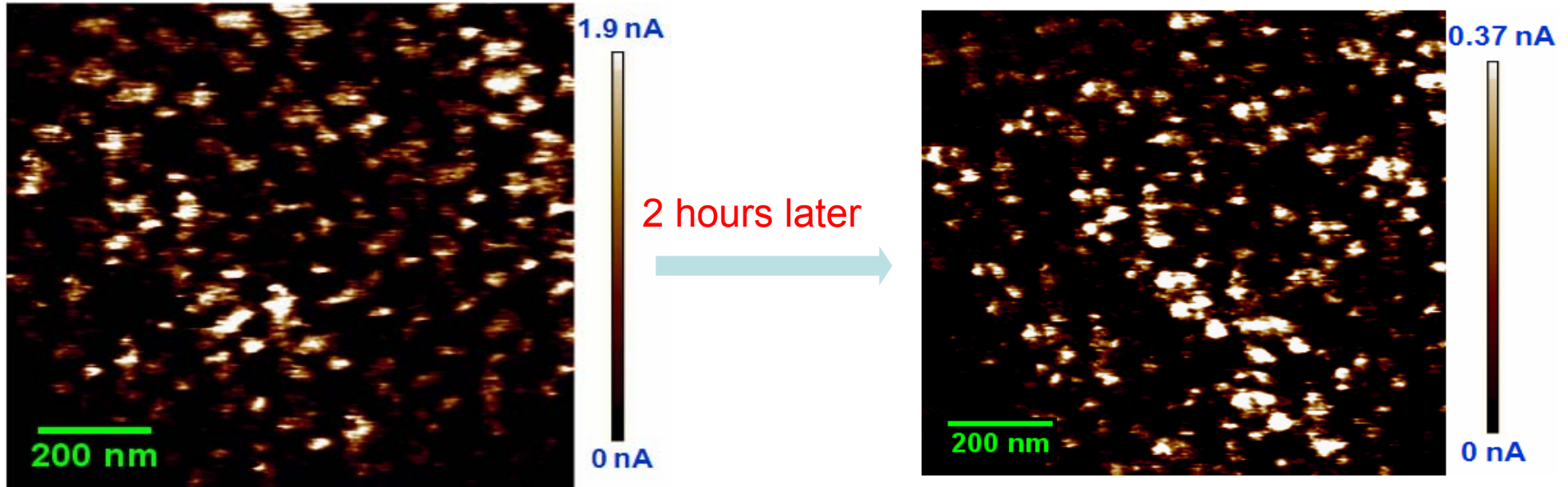
t = thickness of deposited Ge layer

## SEM image



# NEAR-FIELD EBIC

## Memory effect



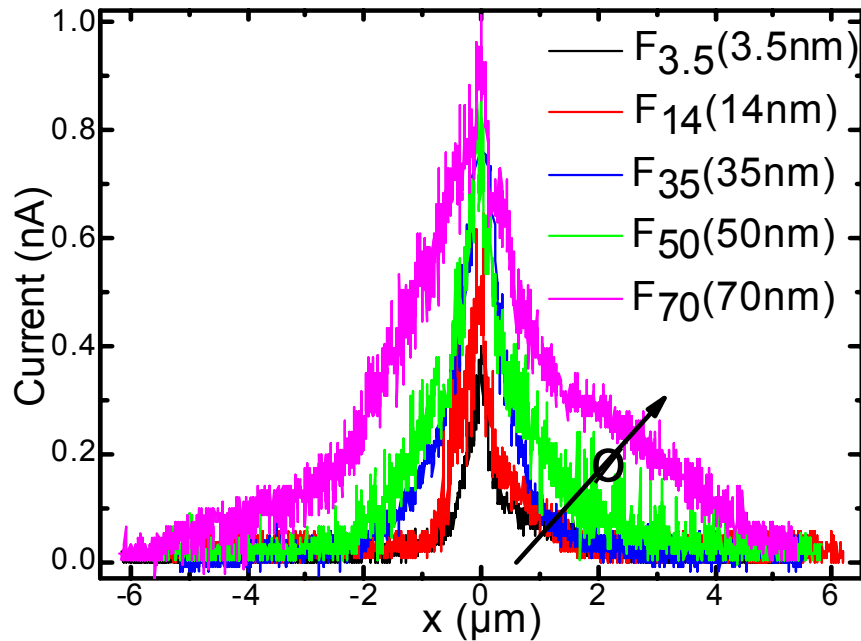
NF-EBIC (Ebeam = 5keV,  $V = 0V$ )

C-AFM (beam off,  $V = 0V$ )

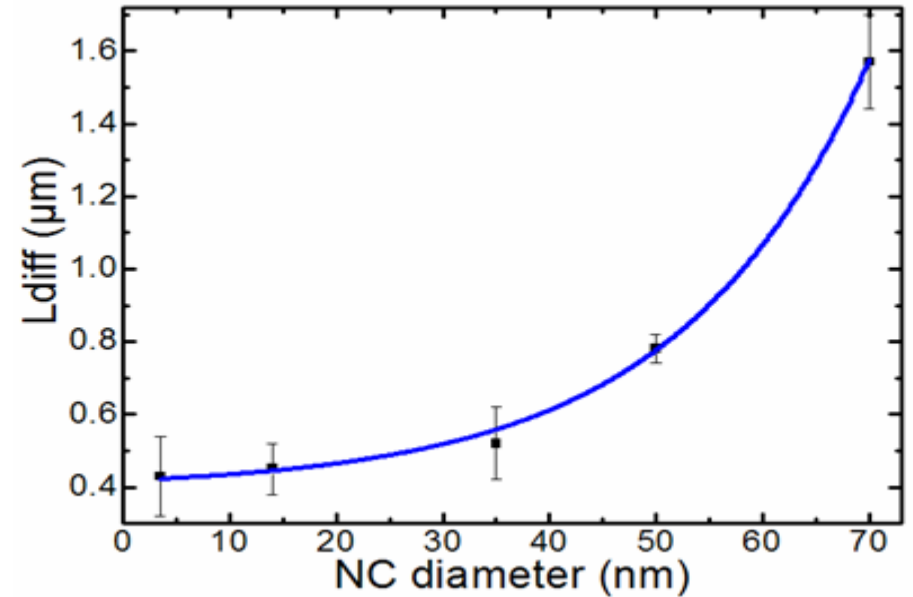


Evidence of memory effect in the nanocrystals

# Diffusion length variation with NC diameter



Current profiles



Exponential variation of  $L_{\text{diff}}$  v.s NC diameter

The increase of diffusion length with the NC size is attributed to a stronger potential overlapping when the NC size increases, resulting in a lateral carrier exchange between the QDs.

# CONCLUSION

- The resolution of EBIC analysis can be strongly improved by combining AFM and SEM
- The resolution of near-field EBIC or nano-EBIC is limited by the size of the depletion layer created by AFM tip-surface contact, not by the generation volume
- We have seen through several examples that resolution is around 20 nm

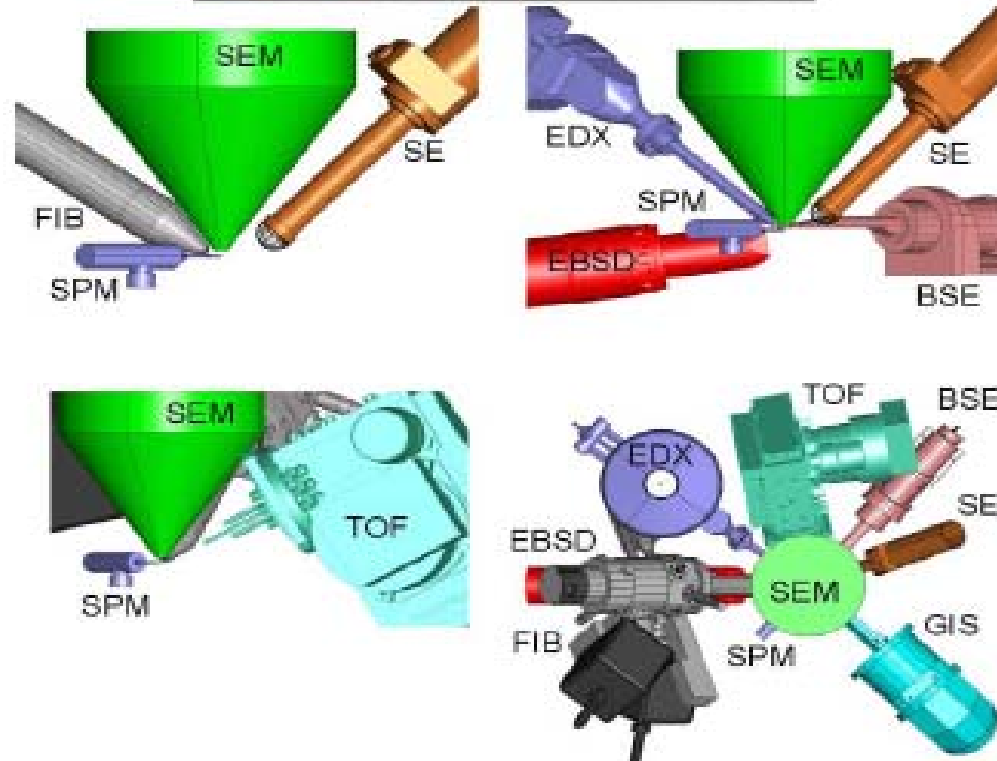
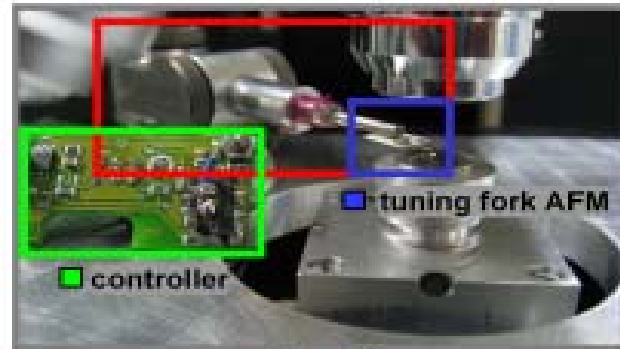
## Acknowledgements

- G. Saint-Girons and A. Lemaître from the laboratory of Photonics and Nanostructures (Marcoussis, France)
- The Group of I. Berbézier, IM2NP, CNRS, Marseille, France



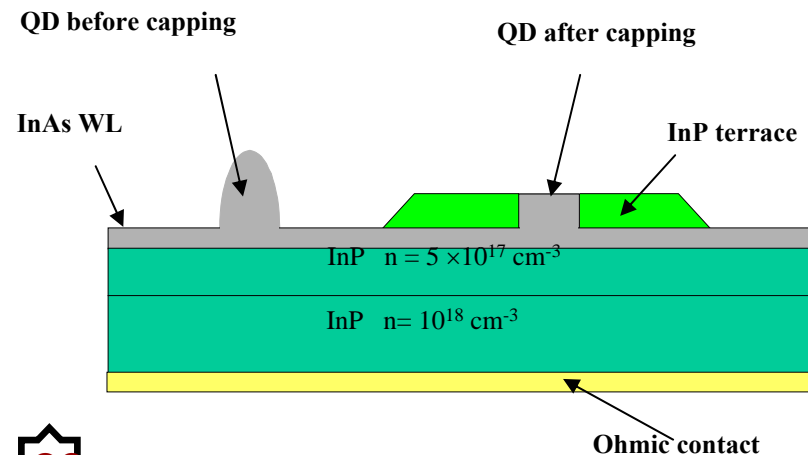
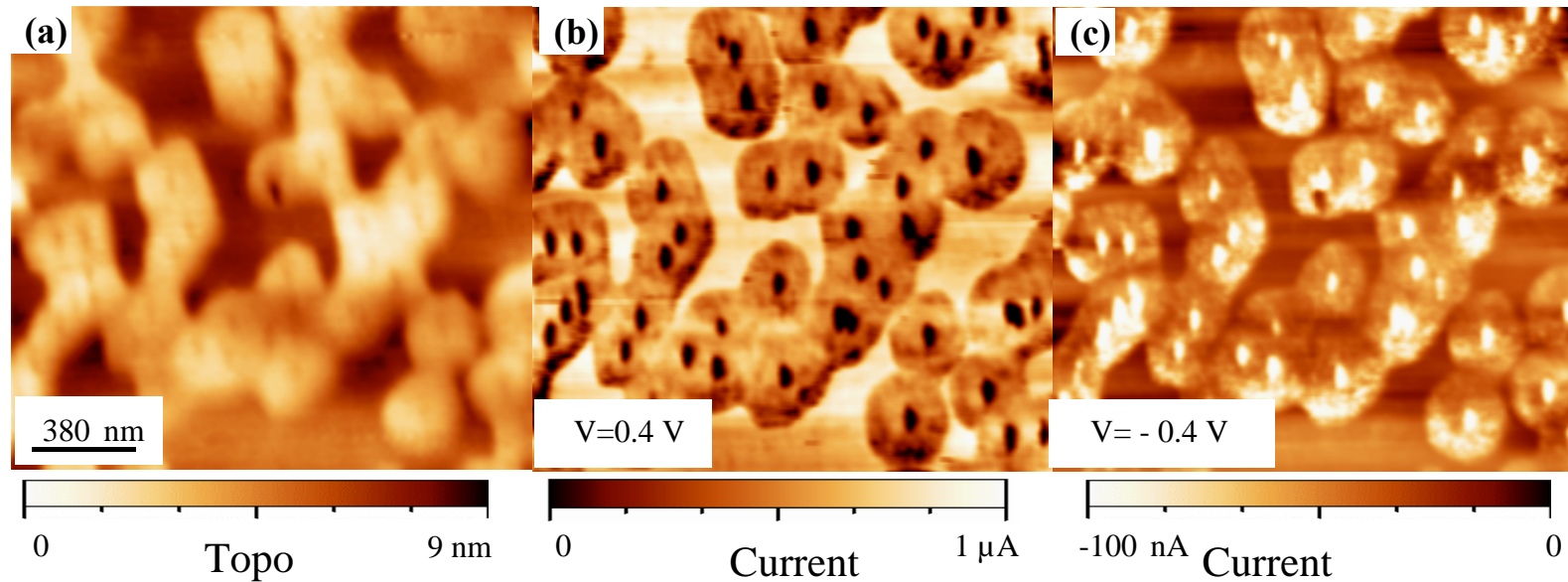
This work was supported in part by the European Community FP7 program in the scope of the FIBLYS project under grant CP-TP 214042-2.

# Projet 7ème PCRD : « building an analyzing focused ion beam for nanotechnology »



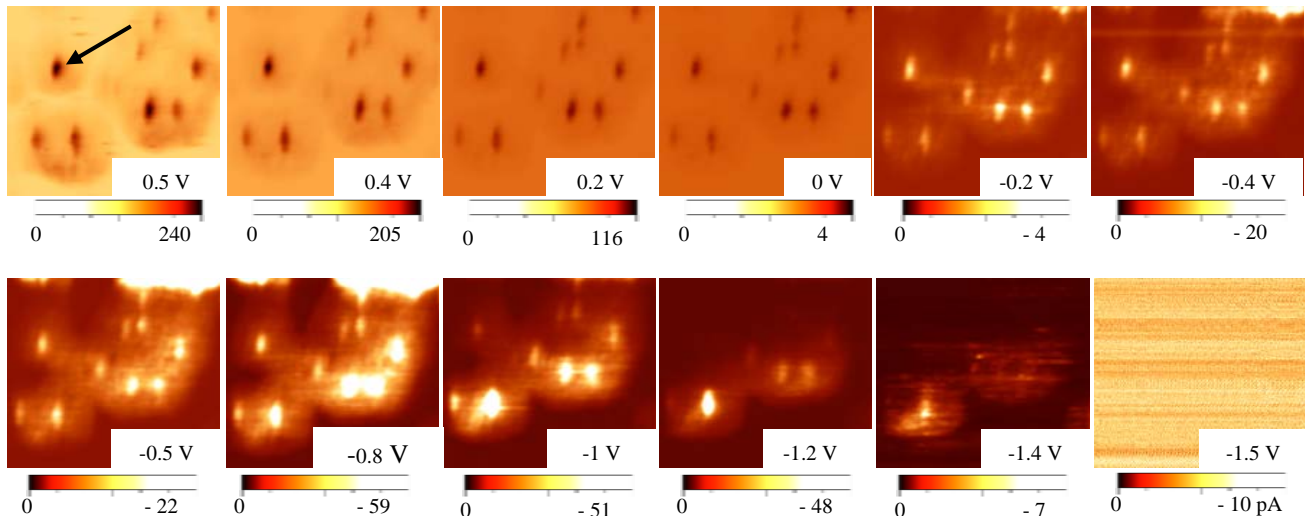
# Conductive-AFM

InAs QDs grown on InP(001) by MOVPE and covered with a 5 nm thick layer

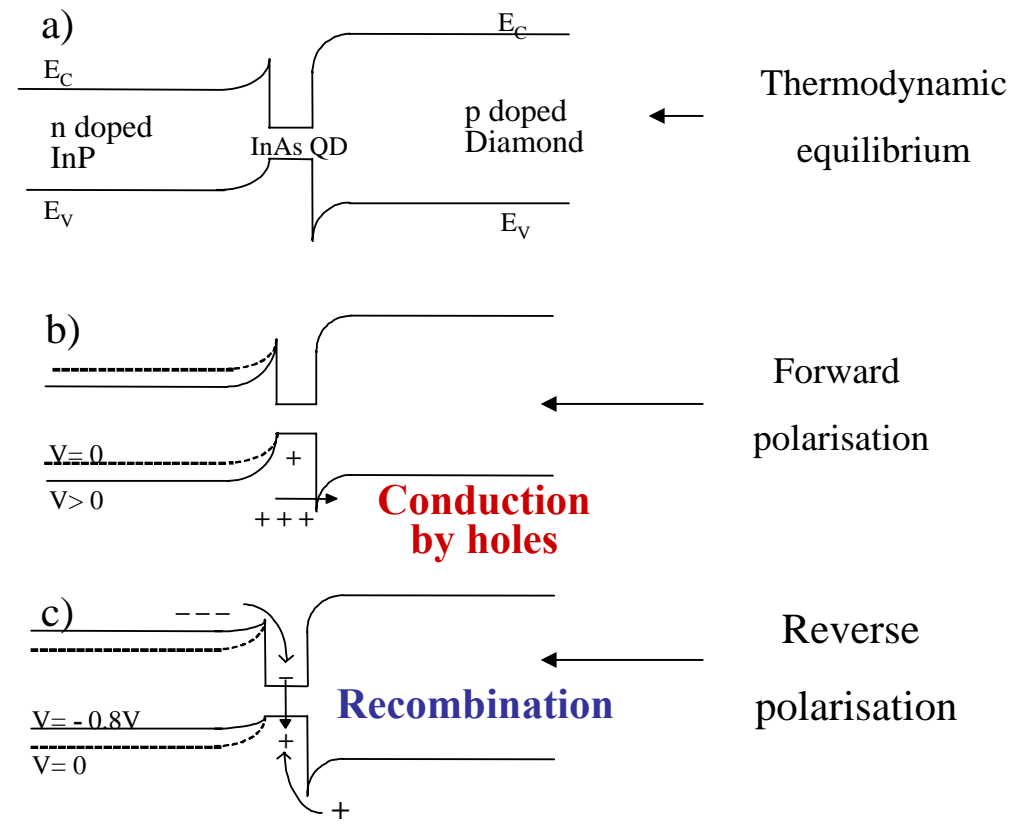
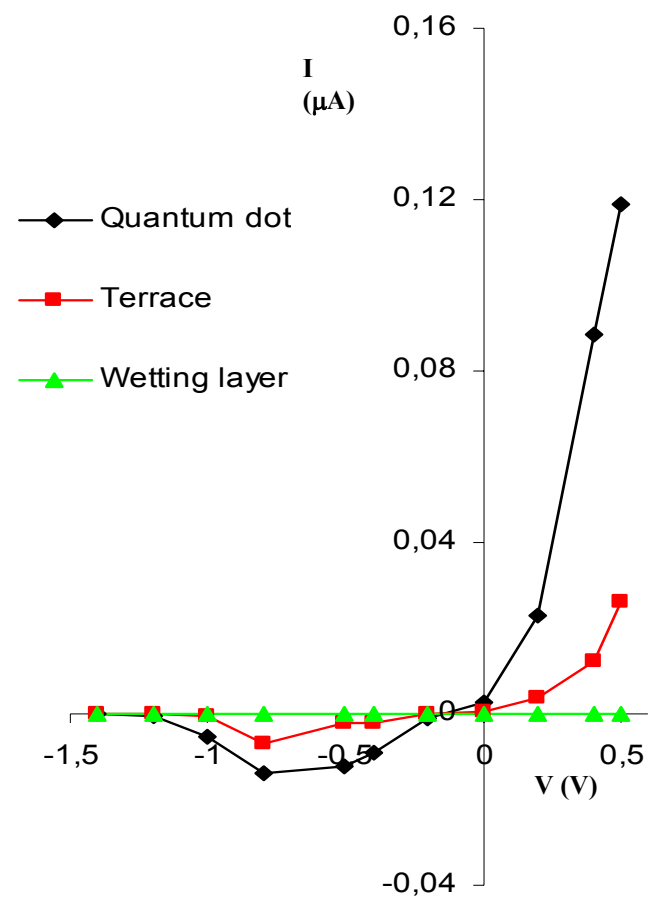


- The cap-layer does not entirely cover the surface but forms terraces around the dots
- 3 levels of current:
  - Very high on the dots
  - $\sim 10$  times less on the terrace
  - Null on the wetting layer

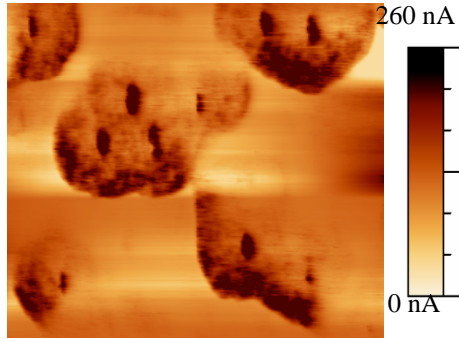
# Electronic transport



## I-V characteristics

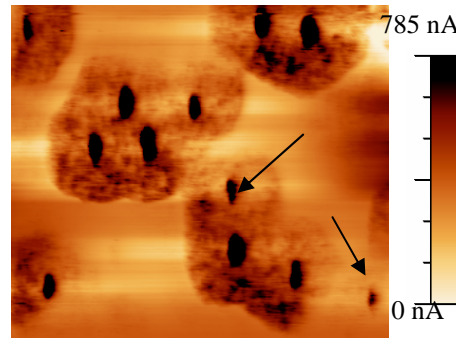


# Writing-erasing



V=0.3 V

1st image



V=0.3 V

2nd image

**Current increases in the 2nd image**

# Memory effect

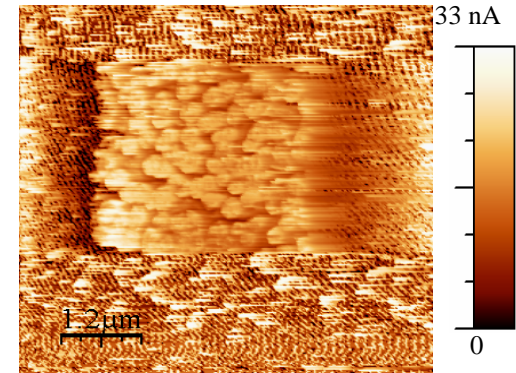


Image taken at 0.3 V

one hour after the

Central part at 0.8 V

**QDs and terraces trap charges**

**Discharging occurs for negative bias**